2	Fuel-free nanocap-like motors actuated under visible light
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16	Brownian motion.
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1 Abstract

The motion of nanomotors triggered by light sources will provide new alternative routes to 2 power nanoarchitectures without the need of chemical fuels. However, most light-driven 3 nanomotors are triggered by UV-light, near infrared reflection (NIR) or laser sources. We 4 demonstrate that nanocap shaped Au/TiO₂ nanomotors (175 nm in diameter) display increased 5 Brownian motion in the presence of broad spectrum visible light. The motion results from the 6 surface plasmon resonance (SPR) effect leading to self-electrophoresis between the Au and TiO₂ 7 layers, a mechanism called plasmonic photocatalytic effect in the field of photocatalysis. This 8 mechanism has been experimentally characterized by electron energy loss spectroscopy (EELS), 9 10 energy-filtered transmission electron microscopy (EFTEM) and optical video tracking. We also 11 studied this mechanism in a more theoretical manner using numerical finite-difference time-12 domain (FDTD) simulations. The ability to power nanomaterials with visible light may result in entirely new applications for externally powered micro/nanomotors. 13

14 Introduction

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Emerging autonomous nano/micromotors are promising man-made devices that could 16 eventually be used in both biomedical and environmental applications.^[1] Early research focused on 17 the investigation of catalytic motors, which need chemical components as propellants.^[2] The 18 intrinsic biotoxicity of the most widely used fuel, H₂O₂,^[3] hinders a much broader application in 19 biomedicine. Although recent reports demonstrated that bio-friendly fuels such as glucose and urea 20 could power nanomotors,^[4] the sustained fuel-consumption severely limits the lifetime of such 21 motors. Therefore, external stimuli, including ultrasound,^[5] magnetic^[6] and electric fields,^[7] have 22 been used as fuel-free and biofriendly power sources. 23

Visible light is becoming an attractive external stimulus to propel nano/micro objects because it 1 2 represents a ubiquitous and theoretically unlimited fuel source that is non-intrusive, clean, and easily controllable. Most studies to date focused on nano/micromotors powered by non-3 biocompatible UV light^[8] or not easily accessible lasers.^[9] To overcome these challenges, a range 4 of nano/micro objects which can use visible light^[3b, 10] have been explored. Currently, the reported 5 fuel-free light-driven micromotors rely on two mechanisms: thermophoresis^[11] and 6 thermocapillary effects.^[12] Surface plasmon resonance (SPR) effects have been developed for 7 photocatalysis applications under visible or UV light,^[13] such as photodegradation^[14] and 8 photosplitting.^[15] In these systems, photocatalytic materials with large band gaps, like TiO₂ (~3.20 9 eV), are combined with a plasmonic material (gold) to enhance electron transfer in the visible range. 10 Researchers have demonstrated that the catalytic process can be enhanced by charge transfer due 11 to Plasmon induced photocatalytic effects.^[16] 12

13 We present Au/TiO₂ nanocap motors that display enhanced Brownian motion under visible light 14 through the plasmonic photocatalytic effect. The experimental and simulated localized surface plasmon resonance (LSPR) effect in the Au/TiO₂ nanostructure,^[2b, 15b, 16a] coupled with the electric 15 16 field distribution and thermal diffusion of electrons, induce enhanced Brownian motion of nanocap motors. Materials characterization, theoretical simulations, video tracking analysis of nanomotors 17 18 and control experiments are presented to confirm the motion of nanocaps in a fuel-free manner. These interpretations are consistent with simulations by Crozier et al.,^[17] Yuan et al.^[18] and Baffou 19 et al.^[19] of metallic nanoparticles containing Au and TiO₂ materials. 20

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1 Results and Discussion

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3 The Janus nanocaps were fabricated by combining plasma etching and physical vapor deposition methods (Figure 1a). Monolayers of 200 nm polystyrene particles (PS NPs) were prepared by drop 4 casting method and followed by Ar plasma etching to separate the packed PS NPs with nanogaps. 5 Figure panels 1b and c show the obtained monodispersed monolayers of PS NPs.^[20] The Au and 6 7 TiO₂ layers were successively evaporated on the loosely packed PS NPs patterns to form Janus PS/Au/ TiO₂ particles by electron beam deposition. The obtained Janus nanoparticles were 8 annealed at 600 °C for 1 hour to produce anatase TiO₂ layers. Finally, the removal of PS template 9 by high temperature evaporation results in the nanocap structure (Figure 1d and e). X-Ray 10 diffraction (XRD) measurements show that the annealed nanocaps mainly contain anatase TiO2 11 12 (Figure S1a), which provides a good photo-catalytic performance, while the unannealed nanocaps only contain amorphous TiO₂ (Figure S1b). A scanning electron microscopy (SEM) image of the 13 TiO₂ surface layer of the nanocaps is shown in Figure 1f. 14

15 Additionally, we characterized the nanocaps by using electron energy loss spectroscopy (EELS) 16 and transmission electron microscopy (TEM). EELS provides information about the elemental composition of nanocaps and is capable of identifying low atomic number materials such as O in 17 18 the TiO₂ layers. The outer TiO₂ layer and the inner Au layer formed separately without mixing (Figure 2a). In addition, the elemental mapping clearly shows that no elements other than Ti, O, 19 20 and Au were present in the structure. This clear separation of TiO_2 and Au was confirmed from 21 TEM images of nanocaps (Figure 2b). High resolution transmission electron microscopy (HRTEM) provides evidence of the crystalline structure of the interface between TiO₂ and Au (Figure 2c) 22 23 with lattice spacings in the anatase TiO_2 layer of 0.17 nm, 0.23 nm, and 0.163 nm of the (1 0 5), (1 1 2), and (2 1 1) planes, respectively. 24 4

1	In order to demonstrate the Plasmon induced photocatalytic effect, we measure the plasmon
2	resonances of a single Au/TiO_2 nanocap with EELS and energy-filtered transmission electron
3	microscropy (EFTEM). EELS is among the pioneering methods to map plasmons at nanometer
4	scales. ^[21] EELS is often assumed to to be able to reflect the parameter of photonic local density of
5	states (LDOS) ^[22] projected along the trajectory of the electron beam. Meanwhile, EFTEM is used
6	to show the spatial distribution of a selected plasmon resonance in a single Au/TiO $_2$ nanocap. A
7	resonant plasmon peak at 1.75eV (Figure 3a), at both the Au/TiO_2 and TiO_2/vacuum interfaces
8	(Figure 3a), indicates strong absorption around the 708 nm wavelength. Figure 3b presents the
9	spatial distribution of the surface plasmon resonance at 1.2-2.4 eV. The localization areas with
10	higher intensity indicate stronger absorption for that specific energy range.

11 The calculated electron-energy-loss probability (Figure 3c) could confirm the experimental finding that plasmon resonances at both interfaces have similar energies. However, a slight energy 12 13 shift of 1.48 eV in the calculated EEL spectra had been expected because the actual dielectric function of the TiO₂ layer and the orientation of the nanocap^[23] cannot be exactly determined. The 14 15 dielectric function of TiO₂ strongly depends on its crystal structure. By means of a numerical FDTD simulation, we could show that this nanocap structure sustained a typical dipolar plasmon mode 16 (Figure 3d). Furthermore, the total electric field of the exited surface plasmons (E_{tot}) is mostly 17 18 localized at the outer edge between TiO₂/Vacuum and Au/TiO₂, as well as at the Si₃N₄ substrate below the nanocap/substrate interface. It should be noticed that optical excitation can introduce 19 pressure on the structure. The density of the induced electromagnetic momentum and the stress 20 tensor both depend on the electric field and the magnetic field distributions. Symmetries of the 21 different field components however are not the same; This effect introduces an additional 22

asymmetry to the induced pressure on the particle. Thus the optical pressure manipulates the
 motion of the particle dissimilar to the mechanism of heat transfer.

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To investigate the influence of this plasmon effect on self-propulsion, a white halogen cold light 4 source (LCD lamp) with an intensity of 100 mW/cm² was used to illuminate the samples. Surface 5 plasmon resonance on the Au surface, due to the incident LCD light, excites the free electrons on 6 7 the metal surface, causing the nanocaps to appear as bright spots on microscope. The higher light 8 intensity means more charges are in oscillation (i.e., excited) on the Au surface, which in turn leads 9 to a more pronounced Plasmon induced photocatalytic effect. Higher light intensities lead to an 10 increased proton gradient and fluid shear velocity, due to the increased fluid flow around the particle caused by higher charge separation (Figure 4a). The light absorption spectrum for the 11 Au/TiO₂ nanocaps showed an absorption peak at 690 nm (Figure 4b). 12

The plasmonic photocatalytic mechanism results is an increased Brownian motion of the 13 14 nanocaps upon visible light. The tracking trajectories of representative light-activated nanocaps tracked for 10s are displayed in Figure 4c. In order to observe the nanocaps, we had to use the 15 intensity microscope focus light to illuminate the sample. This means that our "dark" control or 16 baseline (Fig. 4d, e, f) was in fact illuminated with the microscope's 1 mW/cm² focus light. The 17 diffusion coefficient (D) is given by $D=MSD/i \cdot \Delta t$ and for the case of a two-dimensional Brownian 18 motion analysis, i is equal to 4. The MSD values (in 1 second interval time) for these low 19 (microscope only) and high (microscope plus LCD) light intensities respectively are plotted in 20 21 Figure 4d. We also tracked 12 Au/TiO₂ nanocaps to investigate how turning LCD light ON and OFF would affect their mobility. The diffusivity of an individual nanocap decreased from $1.87 \pm$ 22 $0.5 \,\mu m^2/s$ (LCD on) to $1.04 \pm 0.2 \,\mu m^2/s$ when the LCD light was turned off (Figure 4e). These results 23

Comentado [SS1]: For how long is the tracking?

provide direct proof of the correlation between increased Brownian motion of nanocap motors and
 visible light intensity.

3 Within the plasmonic photocatalytic mechanism, there are two possible contributions that may 4 affect Brownian motion of the nanocaps: (1) electron diffusion between the Au and TiO_2 layers due to electrons in the Au layer becoming excited by the incident visible light, and (2) dissipation of 5 part of the incident light energy as heat and its diffusion into the surrounding fluid.^[16a, 16b, 24] In the 6 first mechanism, some separated electrons are lost through recombination and other surface losses, 7 including thermal diffusion of the excited electrons on the Au side, while the remaining electrons 8 are injected into the conduction band of TiO2. This charge separation creates an electric field 9 around the nanocap. The generated protons from the oxidation of water at the Au layer flow to the 10 surface of TiO₂ and are then reduced by diffused electrons at TiO₂ layer.^[8b, 25] The diffusion of 11 proton products due to electron flow between the two layers, can induce a solute flow on the 12 surrounding fluid from the Au inner surface to the TiO₂ outer surface via self-electrophoresis 13 14 (Figure 4a). This provides a propelling force to move the nanocaps forward. Self-electrophoresis is a process in which the charged microparticles move in a self-generated electric field due to the 15 16 asymmetric distribution of ions. In previous reports on visible light activated Au/TiO₂ microparticles,^[25-26] the electrons produced from the Au layer were transported to and consumed 17 by the exposed TiO₂ surface^[26] resulting in a fluid flow towards TiO₂ and thus propulsion of the 18 microparticles. Hong et al.^[8b] showed that anatase TiO₂ particles are very active and can move 19 20 under UV light. The XRD spectrum (Figure S 1a) shows the formation of a uniform anatase phase. 21 Visible light cause excitation of large amounts of free electrons on the Au layer due to the 22 Plasmon induced photocatalytic effect. This brings about a negative shift in the Fermi energy level of TiO₂. In order to investigate the motion mechanism, we performed three control experiments 23

1	(Figure 4f). First, the Au layer was replaced by 20 nm Pt layer to form Pt/TiO_2 nanocaps. Pt
2	nanolayer is not as active as Au when light is incident on it as showed by the weak visible light
3	absorption in Figure S2. Without LCD light, the diffusion coefficient of Pt/TiO_2 nanocaps is 1.31
4	\pm 0.36 $\mu m^2/s.$ When light is turned on, their diffusion coefficient slightly increases to 1.45 \pm 0.25
5	$\mu m^2/s$ (Figure 4f). Pt does not show any significant peak in absorption of visible light, thus causing
6	a shallow Plasmon resonance which $% 100000000000000000000000000000000000$
7	and the fact that there is no increase in this diffusion after LCD light ilumination. This experiment
8	indicates that Au as active material, with high absorption visible light band is crucial for the
9	enhanced Brownian motion of the nanocaps via Plasmon photocatalytic effect.

10 In another control experiment, we synthesized nanocaps composed of 20 nm Au, 10 nm SiO₂, and 25 nm TiO₂ layers. Here, the SiO₂ sandwiched layer separates the direct contact between Au 11 and TiO2 blocking the flow of electrons, which leads to a blue shift compared to the UV-vis 12 absorption peak of 20 nm Au/25 nm TiO2 nanocaps (Figure S2). When LCD light is on, the 13 diffusion coefficient value is increased slightly from $0.84 \pm 0.26 \,\mu$ m²/s to $0.9 \pm 0.24 \,\mu$ m²/s (Figure 14 4f) with no significant differences, which is much lower than the values obtained with the TiO_2/Au 15 16 nanomotors. These two controls prove the effect of Plasmon induced electron flow in the TiO₂/Au 17 nanomotors.

The second contributor to the motion relies on the rapid heat transfer away from the water-Au interface aiding the nanocap movement through thermophoresis. An average temperature increase of $0.2 \pm 0.05^{\circ}$ C was detected in the bulk solution (Figure S4) when LCD light was turned on the nanocaps. We fabricated Au/SiO₂ nanocaps as the control samples to evaluate the thermophoretic effect of light-driven nanomotors ruling out the Plasmonic photocatalytic effect effect as the SiO₂ layer can prevent hot electrons transport from Au layer. The light absorption spectra of these

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Comentado [SS2]: Pt absorption was too low and produces shallow plasmon

1	modified nanocaps clearly show that $\mathrm{Au}/\mathrm{SiO}_2$ and $\mathrm{Au}/\mathrm{amorphous}\ \mathrm{TiO}_2$ nanocaps do not have the
2	ability to absorb visible light (red and blue curves in Figure S2). We measured the diffusion
3	coefficient value of the Au/SiO ₂ nanomotors (Figure 4f) by tracking them under LCD light is 0.98
4	\pm 0.08 $\mu m^2/s,$ which is almost the same as the value without LCD light (0.97 \pm 0.07 $\mu m^2/s)$ with
5	no significant differences. These results indicate that, at this light power and wavelength, the thin
6	Au layer on the nanocaps is not sufficient to provoke an enhanced Brownian motion and hence the
7	thermophoresis is in this case not a significant effect.

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9 Conclusions

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We demonstrated that Au/TiO₂ nanocaps exhibit active Brownian motion when illuminated by 11 12 an LCD light, thus becoming nanomotors that were activated by visible light. The mechanism for the motors' propulsion was interpreted as the fast electron transfer between the two layers of the 13 nanocaps leading to a self-electrophoresis effect under visible light. Both, the experimental and the 14 simulation results of Au/TiO₂ plasmon resonance absorption supported this proposed motion 15 mechanism. It was found that these photoactive nanocaps react to changes in light intensity 16 produced by an LCD lamp when light was turned on and off. This work provides a novel structure 17 and fabrication technique which could be used for the activation and manipulation of biocompatible 18 nano/micro machines, merely requiring visible light as a propulsion mechanism. 19

20 Experimental Section

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- Fabrication of Nanocaps. 175 nm Au/TiO₂ nanocaps were fabricated by electron beam physical
 vapor (E-beam) deposition and annealing. A polystyrene monodispersed particle (Sigma-Aldrich)
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1	monolayer was prepared by drop casting method with 200 nm PS in ethanol solution. Followed by
2	an argon plasma etching process (60 W radio frequency power with 0.5 mm Hg), $^{[20]}$ PS particles
3	were separated from one another and their sizes were reduced to 130 nm . Then, 20 nm Au and 30
4	nm TiO ₂ layers from pure Au and TiO ₂ pellets (Kurt J. Lesker Company) were deposited on the
5	upper surface of the PS monolayer. After annealing for 1h at 600 $^\circ\text{C},$ amorphous TiO_2 was turned
6	to Anatase TiO_2 and PS templates were removed. Finally, Au/TiO ₂ nanocaps on silicon wafer were
7	collected with DI water by 30 min ultrasound sonication. 175 nm Pt/TiO_2 nanocaps were prepared
8	with the same method by using Pt pellets instead of Au. For 185 nm Au/SiO_2/TiO_2 nanocaps, 20
9	nm Au, 10 nm SiO ₂ and 35 nm TiO ₂ layers were deposited on 100 nm PS monolayer sequentially
10	before 1 h annealing process in 600 $^{\circ}\mathrm{C}$ oven. 175 nm Au/SiO_2 and Au/amorphous TiO_2 nanocaps
11	were prepared by E-beam deposition with 20 nm Au/25 nm SiO_2 and 20 nm Au/25 nm TiO_2 layers
12	which were then washed with DMF solution to remove PS templates.

13 Instruments

Images of PS particles, Janus structures, and nanocaps were captured by a Zeiss ULTRA 55 SEM. 14 HRTEM and STEM-EELS spectra were recorded in a Jeol ARM 200F-DCOR equipped with a 15 field-emission electron source, a probe Cs corrector, and a Gatan GIF Quantum ERS spectrometer. 16 EFTEM imaging, and EELS measurements were conducted with the Zeiss SESAM microscope. 17 18 The microscope was equipped with an electron monochromator (CEOS Heidelberg) and the MANDOLINE energy filter. Optical videos were recorded by Leica optical microscopy. The KL 19 20 1500 LCD (SCHOTT) close to 100 CRI halogen spectrum and original focus light from Leica 21 optical microscopy were used as illuminators. Testo 85-infrared Thermometer was used to detect 22 the temperature changes of Au/TiO₂ nanocaps. DLS measurements were operated by Malvern 23 Zetasizer Nano ZS.

SPR Peak Absorption and FDTD Simulation 1

- EELS data were acquired with an energy resolution of 90 meV as determined from the full width 2
- at half maximum of the zero-loss peak (ZLP). The acquisition time for each spectrum was 0.8 s. 3
- The energy-loss spectrum was dispersed perpendicular to the energy-dispersive direction on the 4
- 5 CCD camera in order to make the full dynamic range of almost 1000 between ZLP and plasmon
- peaks accessible without saturating the camera. 6
- In order to perform the simulations, a numerical finite-difference time-domain (FDTD) method 7
- was used.^[27] The whole simulation domain was discretized with unit cells of 1.5 nm edge, and a 8
- higher-order absorbing boundary condition has been exploited to meet the far-field radiation 9
- boundary conditions. In order to model the permittivity of both gold and TiO₂, a Drude model in 10
- addition to two critical-point functions has been introduced into the simulation domain.^[28] 11

12 **Author Contributions**

13 X.W., V.S. and S.S. initiated the project. X.W. and V.S. prepared the nanocaps and carried out the experiments. S. G., N.T. and P.v.A. provided the plasmon-related experiment and simulation 14 results. K.H. conducted the TEM imaging and EELS mapping. A. M.L. developed a code for 15 tracking nanocaps and analyzed their motion. All the authors contributed to the paper writing. 16 Author 1 and Author 2 contributed equally to this work. 17

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1 Figures

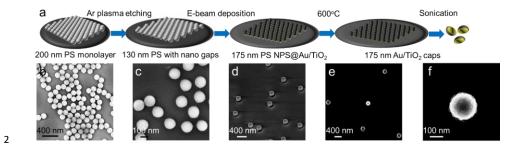


Figure 1. Fabrication of nanocaps. (a) Schematic of the multi-step fabrication procedure of 175
nm Au/TiO₂ nanocaps. (b-f) Scanning electron microscope images of (b) 200 nm polystyrene
particles (PS), (c) loosely packed PS film after Ar plasma etching, (d) Janus PS/Au/TiO₂ particles
after E-beam deposition, (e) Au/TiO₂ nanocaps without PS after annealing process, and (f) a single
Au/TiO₂ nanocap.

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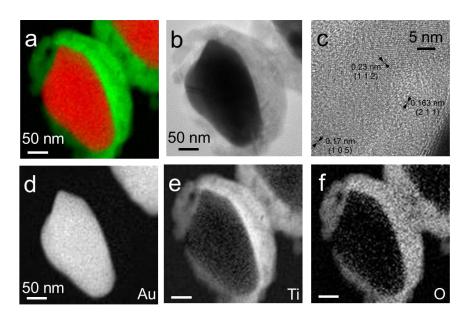


Figure 2. Characterization of nanocaps motors. (a) color-coded elemental map of Au (red, cf. panel
d) and Ti (green, cf. panel e). (b) HRTEM image of the 175 nm nanocap with the 20 nm Au core
layer shown as dark grey Au surrounded by the 25 nm TiO₂ shell layer (light grey). (c) HRTEM
image of the interface between Au and the TiO₂ layer at high magnification. (d–f) EELS spectrum
images of (d) Au-M_{4,5} at 2206 eV, (e) Ti-L_{2,3} at 455 eV, and (f) O-K at 532 eV.

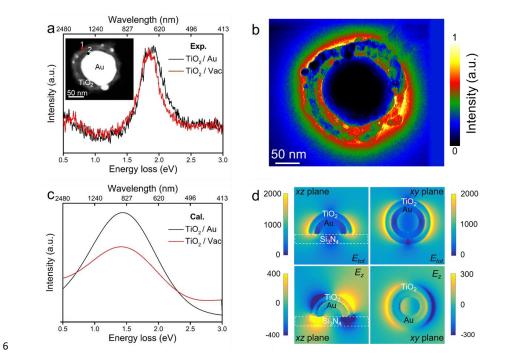
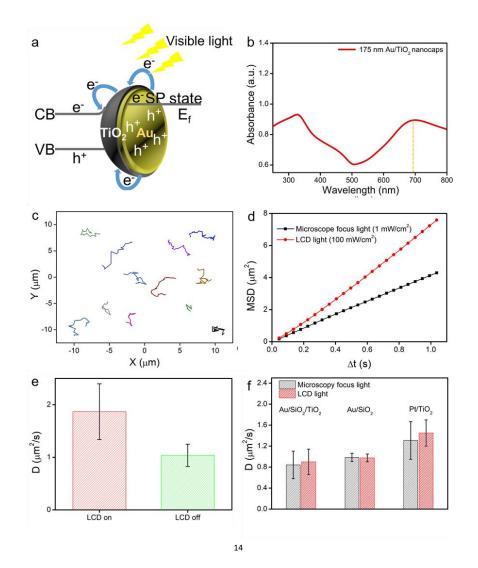


Figure 3. (a) Experimental EEL spectra (zero-loss peak corrected) at the interfaces of Au/TiO₂
and TiO₂/Vacuum. The inset shows the STEM-HAADF image of a single Au/TiO₂ nanocap and
the two highlighted spots indicate the considered impact parameters. (b) 1.2-2.4 eV energyfiltered TEM image of a single Au/TiO₂ nanocap. (c) Calculated electron-energy-loss probability
at the interfaces of Au/TiO₂ and TiO₂/Vacuum. (d) Spatial distribution of the calculated total

electric field (color code) $E_{tot.} = \sqrt{|E_x|^2 + |E_y|^2 + |E_z|^2}$ and E_z in the *xz* and *xy* planes (above the substrate) of a single Au/TiO₂ nanocap when the structure is excited with linearly polarized light along the *x*-direction at 1.5 eV. The structure is above a 30 nm-thick Si₃N₄ substrate in FDTD simulations, as in our EELS and EFTEM measurements.



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1	Figure 4. Enhanced Brownian motion of nanocaps. (a) Schematic cartoon for the surface plasmon
2	resonance effect of Au/TiO ₂ under visible light. (b) Light absorption spectrum of 175 nm Au/TiO ₂
3	nanocaps. (c) Trajectory of 12 nanocaps with X and Y coordinates under LCD light tracked for
4	10s. (d) Average MSD versus time interval ($\Delta t=1$ s) for 12 tracked 175 nm nanocaps with and
5	without LCD light (100 mW/cm ²). (e) Diffusion coefficient for 12 Au/TiO_2 nanomotors with cycled
6	LCD light (on/off). Each track corresponds to 8 s. (f) Average diffusion coefficient values of
7	Au/SiO ₂ /TiO ₂ , Au/SiO ₂ , Pt/TiO ₂ nanocaps with focus light (1 mW/cm ²) and LCD light (100
8	mW/cm^2).
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1 References

- a) J. Wang, ACS Nano 2009, 3, 4; b) S. Sengupta, M. E. Ibele, A. Sen, Angewandte Chemie International Edition
 2012, 51, 8434; c) M. Guix, C. C. Mayorga-Martinez, A. Merkoçi, Chemical Reviews 2014, 114, 6285; d) S.
 Sánchez, L. Soler, J. Katuri, Angewandte Chemie International Edition 2015, 54, 1414; e) H. Wang, M.
 Pumera, Chemical Reviews 2015, 115, 8704; f) J. Katuri, X. Ma, M. M. Stanton, S. Sánchez, Accounts of
 chemical research 2016, 50, 2; g) F. Wong, K. K. Dey, A. Sen, Annual Review of Materials Research 2016, 46,
 407; h) P. Illien, R. Golestanian, A. Sen, Chemical Society Reviews 2017.
- 8 [2] a) R. F. Ismagilov, A. Schwartz, N. Bowden, G. M. Whitesides, Angewandte Chemie International Edition
 9 2002, 41, 652; b) V. Subramanian, E. E. Wolf, P. V. Kamat, Journal of the American Chemical Society 2004,
 126, 4943; c) H. Wang, G. Zhao, M. Pumera, Journal of the American Chemical Society 2014, 136, 2719.
- [3] a) Y. Li, F. Mou, C. Chen, M. You, Y. Yin, L. Xu, J. Guan, *RSC Advances* 2016, 6, 10697; b) D. Zhou, Y. C. Li, P.
 Xu, N. S. McCool, L. Li, W. Wang, T. E. Mallouk, *Nanoscale* 2017, 9, 75.
- 13
 [4]
 a) D. Pantarotto, W. R. Browne, B. L. Feringa, Chemical Communications 2008, DOI: 10.1039/B715310D1533;

 14
 b) X. Ma, A. C. Hortelao, A. Miguel-López, S. Sánchez, Journal of the American Chemical Society 2016, DOI:

 15
 10.1021/jacs.6b06857; c) X. Ma, X. Wang, K. Hahn, S. Sánchez, ACS Nano 2016, 10, 3597.
- [5] a) W. Wang, L. A. Castro, M. Hoyos, T. E. Mallouk, ACS Nano 2012, 6, 6122; b) V. Garcia-Gradilla, S.
 Sattayasamitsathit, F. Soto, F. Kuralay, C. Yardimci, D. Wiitala, M. Galarnyk, J. Wang, Small 2014, 10, 4154.
- 18 [6] a) P. Fischer, A. Ghosh, Nanoscale 2011, 3, 557; b) L. Baraban, R. Streubel, D. Makarov, L. Han, D. 19 Karnaushenko, O. G. Schmidt, G. Cuniberti, ACS nano 2013, 7, 1360; c) F. Qiu, S. Fujita, R. Mhanna, L. Zhang, 20 B. R. Simona, B. J. Nelson, Advanced Functional Materials 2015, 25, 1666; d) B. Jang, E. Gutman, N. Stucki, 21 B. F. Seitz, P. D. Wendel-García, T. Newton, J. Pokki, O. Ergeneman, S. Pané, Y. Or, Nano letters 2015, 15, 22 4829; e) S. Kim, S. Lee, J. Lee, B. J. Nelson, L. Zhang, H. Choi, Scientific reports 2016, 6, 30713; f) X. Yan, Q. 23 Zhou, M. Vincent, Y. Deng, J. Yu, J. Xu, T. Xu, T. Tang, L. Bian, Y.-X. J. Wang, K. Kostarelos, L. Zhang, Science 24 Robotics 2017, 2; g) X. Z. Chen, M. Hoop, N. Shamsudhin, T. Huang, B. Ozkale, Q. Li, E. Siringil, F. Mushtaq, L. 25 Di Tizio, B. J. Nelson, S. Pane, Adv Mater 2017, 29; h) T. Li, J. Li, K. I. Morozov, Z. Wu, T. Xu, I. Rozen, A. M. 26 Leshansky, L. Li, J. Wang, Nano Letters 2017, 17, 5092.
- 27 [7] G. Loget, A. Kuhn, Nat Commun 2011, 2, 535.
- 28 [8] a) M. Ibele, T. E. Mallouk, A. Sen, Angewandte Chemie International Edition 2009, 48, 3308; b) Y. Hong, M. 29 Diaz, U. M. Córdova-Figueroa, A. Sen, Advanced Functional Materials 2010, 20, 1568; c) B. Dai, J. Wang, Z. 30 Xiong, X. Zhan, W. Dai, C.-C. Li, S.-P. Feng, J. Tang, Nature nanotechnology 2016, 11, 1087; d) M. Enachi, M. 31 Guix, V. Postolache, V. Ciobanu, V. M. Fomin, O. G. Schmidt, I. Tiginyanu, Small 2016, 12, 5497; e) W. Li, X. 32 Wu, H. Qin, Z. Zhao, H. Liu, Advanced Functional Materials 2016, 26, 3164; f) C. R. Chen, F. Z. Mou, L. L. Xu, 33 S. F. Wang, J. G. Guan, Z. P. Feng, Q. W. Wang, L. Kong, W. Li, J. Wang, Q. J. Zhang, Advanced Materials 2017, 34 29; g) J. Simmchen, A. Baeza, A. Miguel - Lopez, M. M. Stanton, M. Vallet - Regi, D. Ruiz - Molina, S. 35 Sánchez. ChemNanoMat 2017. 3. 65.
- a) B. Qian, D. Montiel, A. Bregulla, F. Cichos, H. Yang, *Chemical Science* 2013, 4, 1420; b) A. P. Bregulla, H.
 Yang, F. Cichos, *ACS Nano* 2014, 8, 6542; c) Z. Wu, X. Lin, Y. Wu, T. Si, J. Sun, Q. He, *ACS Nano* 2014, 8, 6097;
 d) J. Shao, M. Xuan, L. Dai, T. Si, J. Li, Q. He, *Angewandte Chemie International Edition* 2015, 54, 12782; e)
 M. Xuan, Z. Wu, J. Shao, L. Dai, T. Si, Q. He, *J Am Chem Soc* 2016, 138, 6492.
- a) M. J. Esplandiu, A. Afshar Farniya, A. Bachtold, ACS nano 2015, 9, 11234; b) R. Dong, Y. Hu, Y. Wu, W. Gao,
 B. Ren, Q. Wang, Y. Cai, Journal of the American Chemical Society 2017, 139, 1722; c) B. Jang, A. Hong, H. E.
 Kang, C. Alcantara, S. Charreyron, F. Mushtaq, E. Pellicer, R. Büchel, J. Sort, S. S. Lee, B. J. Nelson, S. Pané,
 ACS Nano 2017, 11, 6146; d) J. Wang, Z. Xiong, X. Zhan, B. Dai, J. Zheng, J. Liu, J. Tang, Adv Mater 2017, 29.
- a) H.-R. Jiang, N. Yoshinaga, M. Sano, *Physical Review Letters* 2010, 105, 268302; b) M. Yang, M. Ripoll, *Soft Matter* 2014, 10, 1006.
- 46 [12] C. Maggi, F. Saglimbeni, M. Dipalo, F. De Angelis, R. Di Leonardo, Nat Commun 2015, 6.
- [13] a) S. Linic, P. Christopher, D. B. Ingram, Nat Mater 2011, 10, 911; b) W. Hou, S. B. Cronin, Advanced
 Functional Materials 2013, 23, 1612.
- [14] K. Awazu, M. Fujimaki, C. Rockstuhl, J. Tominaga, H. Murakami, Y. Ohki, N. Yoshida, T. Watanabe, *Journal of the American Chemical Society* 2008, 130, 1676.

- 1
 [15]
 a) Z. Liu, W. Hou, P. Pavaskar, M. Aykol, S. B. Cronin, Nano Letters 2011, 11, 1111; b) Z. W. Seh, S. Liu, M.

 2
 Low, S.-Y. Zhang, Z. Liu, A. Mlayah, M.-Y. Han, Advanced Materials 2012, 24, 2310; c) S. C. Warren, E.

 3
 Thimsen, Energy & Environmental Science 2012, 5, 5133; d) X.-C. Ma, Y. Dai, L. Yu, B.-B. Huang, Light: Science

 4
 & Applications 2016, 5, e16017.
- [16] a) Y. Tian, T. Tatsuma, *Journal of the American Chemical Society* 2005, 127, 7632; b) A. Furube, L. Du, K. Hara,
 R. Katoh, M. Tachiya, *Journal of the American Chemical Society* 2007, 129, 14852; c) S. Linic, U. Aslam, C.
 Boerigter, M. Morabito, *Nat Mater* 2015, 14, 567.
- [17] a) K. Wang, E. Schonbrun, K. B. Crozier, *Nano Letters* 2009, 9, 2623; b) K. Wang, E. Schonbrun, P. Steinvurzel,
 K. B. Crozier, *Nat Commun* 2011, 2, 469.
- 10 [18] C. Min, Z. Shen, J. Shen, Y. Zhang, H. Fang, G. Yuan, L. Du, S. Zhu, T. Lei, X. Yuan, *Nat Commun* **2013**, 4.
- 11 [19] G. Baffou, C. Girard, R. Quidant, *Physical Review Letters* **2010**, 104, 136805.
- 12 [20] L. Yan, K. Wang, J. Wu, L. Ye, *The Journal of Physical Chemistry B* 2006, 110, 11241.
- 13 [21] M. Kociak, O. Stephan, Chemical Society Reviews 2014, 43, 3865.
- 14 [22] F. J. García de Abajo, M. Kociak, *Physical Review Letters* 2008, 100, 106804.
- 15 [23] R. Jiang, F. Qin, Y. Liu, X. Y. Ling, J. Guo, M. Tang, S. Cheng, J. Wang, Advanced Materials **2016**, 28, 6322.
- 16
 [24]
 L. Du, A. Furube, K. Hara, R. Katoh, M. Tachiya, Journal of Photochemistry and Photobiology C:

 17
 Photochemistry Reviews 2013, 15, 21.
- 18 [25] C. Clavero, Nat Photon 2014, 8, 95.
- 19 [26] M. Serra, J. Albero, H. García, *ChemPhysChem* 2015, 16, 1842.
- 20
 [27]
 a) T. Nahid, S. Wilfried, V. Ralf, A. Peter van, New Journal of Physics 2013, 15, 053013; b) T. Nahid, New

 21
 Journal of Physics 2014, 16, 053021.
- 22 [28] N. Talebi, M. Shahabadi, W. Khunsin, R. Vogelgesang, Opt. Express 2012, 20, 1392.

23