Element-Selective Molecular Charge Transport Characteristics of Binuclear Copper(II)-Lanthanide(III) Complexes

Sebastian Schmitz,^[a] Andrew Kovalchuk,^[b] Alejandro Martín-Rodríguez,^[c] Jan van Leusen,^[a] Natalya V. Izarova,^[d] Svenja D. M. Bourone,^[a] Yong Ai,^[b] Eliseo Ruiz,*^[c] Ryan C. Chiechi,*^[b] Paul Kögerler,*^{[a],[d]} and Kirill Yu. Monakhov*^[a]

^[a] Institut für Anorganische Chemie, RWTH Aachen University, Landoltweg 1, 52074 Aachen (Germany).

^[b] Stratingh Institute for Chemistry & Zernike Institute for Advanced Materials, University of Groningen, Nijenborgh 4, Groningen 9747 AG (Netherlands).

^[c] Departament de Química Inorgànica i Orgànica and Institut de Química Teòrica i Computacional, Universitat de Barcelona, Diagonal 645, 08028 Barcelona (Spain).

^[d] Jülich-Aachen Research Alliance (JARA-FIT) and Peter Grünberg Institute (PGI-6), Forschungszentrum Jülich GmbH, Wilhelm-Johnen-Straße, 52425 Jülich (Germany).

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ABSTRACT: A series of isostructural dinuclear 3d-4f complexes, isolated as $[CuLn(L\cdotSMe)_2(OOCMe)_2(NO_3)]$ ·xMeOH (Ln = Gd 1, Tb 2, Dy 3 and Y 4; x = 0.75–1) and comprising one acetate and two thioether-Schiff base (L·SMe⁻) bridging ligands based on 4-(methylthio)aniline and 2-hydroxy-3-methoxybenzaldehyde (HL·SMe = C₁₅H₁₅NO₂S), was synthesized and fully characterized. The magnetic properties of the charge-neutral {CuLn} complexes are dominated by ferromagnetic Cu^{II}–Ln^{III} exchange interactions. Large-area electron transport studies reveal that the average conductivity of robust, self-assembled {CuLn} monolayers on a gold substrate is significantly lower than that of common alkane thiolates. Theoretical calculations of transmission spectra of individual complexes 1 and 4 embedded between two metallic electrodes show that the molecular current–voltage (I– V) characteristics are strongly influenced by electron transport through the Cu centers and thus fully independent on the lanthanide ion, in excellent agreement with the experimental *I–V* data for 1–4. The β -polarized transmission indicated by calculations of 1 and 4 points out their potential as spin filters. In addition, the reactivity of the title compound 1 with Cu^{II} in a square-pyramidal coordination environment toward methanolate and azide was examined, resulting in the formation of a linear trinuclear complex, $[Cu_2Na(L\cdotSMe)_4]NO_3\cdot 3MeOH$ (5), characterized by antiferromagnetic exchange interactions between the two copper ions.

INTRODUCTION

Formation of thin films consisting of magnetic coordination complexes¹ and analysis of their charge transport characteristics with controlled conductance switching defines an important niche in the development of molecular spintronics.²⁻⁶ In such experiments, changing the nature of top and bottom metallic contact electrodes allows us to create specific measurement environments for assessing and modifying the large-area charge- and spin-dependent transport properties of self-assembly monolayers (SAMs). Such electrical measurements can be performed for common molecular tunnel junctions^{7,8} (e.g., Au^{bottom}–SAM–Au^{top}), spin-polarized junctions⁹ (e.g., Au^{bottom}–SAM–ferromagnetic Ni^{top}) or hybrid systems involving a conformal electrode (e.g., Au^{bottom}–SAM–EGaIn^{top} (Eutectic Gallium-Indium).^{10,11} The latter approach offers particularly interesting perspectives for studying large-area tunneling charge transport¹² across magnetic SAMs, comparing their mechanical^{13,14} and electrical properties to those of e.g. widely investigated alkanethiol SAMs.¹⁵ Herein we aim to determine the adsorption characteristics and the main transmission channel of 3d-4f coordination compounds characterized by intrinsically distinct magnetic states that are differently disposed relative to the Fermi levels of the metallic electrodes in the fabricated Au^{bottom}–heterometal complex–EGaIn^{top} junctions. Specifically, we focus on copper–lanthanide systems¹⁶⁻²⁴ that have shown to exhibit structural motifs of varying complexity²⁵ and interesting magnetic and electrical conductivity properties.

We herein report the preparation, magnetochemistry, adsorption characteristics and electrical transport properties of a family of thioether-augmented Schiff base/carboxylate copper(II)-lanthanide(III) complexes of general formula $[CuLn(L\cdotSMe)_2(OOCMe)_2(NO_3)]\cdot xMeOH$ (Ln = Gd 1, Tb 2, Dy



Scheme 1. Synthesis of the tetradentate Schiff base $HL \cdot SMe (= C_{15}H_{15}NO_2S)$ and compounds 1–6.

3 and Y **4**; x = 0.75-1). With an undercoordinated copper center, the ability to modify its coordination geometry upon reaction with smaller ligands was assessed and the resulting trimetallic, lanthanide-free compound $[Cu_2Na(L\cdot SMe)_4]NO_3\cdot 3MeOH$ (5) was analyzed. Both the novel Schiff base HL·SMe and its metal coordination products were characterized using ¹H and ¹³C nuclear magnetic resonance (NMR), infrared (IR) spectroscopy, electrospray ionization mass spectrometry (ESI-MS), thermogravimetric analysis (TGA), and single-crystal X-ray diffraction. The molecular deposition and the formation of thin films of compounds 1-4 on a gold substrate was studied via infrared reflection-absorption spectroscopy (IRRAS), ellipsometry and scanning tunneling microscopy (STM) combined with an EGaIn tip to form molecular junctions. These large-area transport measurements were accompanied by density functional theory (DFT) calculations in order to gain a deeper insight into the conductivity peculiarities at the targeted bottom electrode-heterometal complex-top electrode interfaces.

RESULTS AND DISCUSSION

Synthesis and Stability. Compounds 1–4 were synthesized under aerobic conditions using a two-step synthetic procedure (Scheme 1). The freshly prepared Schiff base HL·SMe was first reacted in methanol under basic conditions, using triethylamine (Et₃N) as base, together with lanthanide nitrate hexahydrates, $Ln(NO_3)_3$ ·6H₂O (Ln = Gd, Tb, and Dy), or yttrium hexahydrate, $Y(NO_3)_3$ ·6H₂O (as diamagnetic analogue) in a 2.0 : 2.3 : 1.0 molar ratio under reflux conditions for 15 minutes. A 1.0 eq. of copper acetate monohydrate (based on Cu), [Cu₂(OOCMe)₄(H₂O)₂], was subsequently added to the resulting clear orange solution that immediately changed to a dark brown color. After stirring under reflux conditions for further 60 minutes the dark brown

solution was filtered off and the filtrate was stored in a capped vial under ambient atmosphere, precipitating the crystalline title compounds $[CuLn(L\cdot SMe)_2(OOCMe)_2(NO_3)]$ ·xMeOH (Ln = Gd 1, Tb 2, Dy 3 and Y 4; x = 0.75-1) in moderate-to-good yields after one day (24.2 % for 4 and 31.3–74.0 % for 1–3). We note that earlier lanthanide precursors (with Ln^{3+} ions larger than Gd^{3+}) did not yield any product precipitation within the time frame described for 1-4 (see Experimental Section). Bigger lanthanides might lead to ligand rearrangement, which stabilizes their coordination more efficiently. (Bigger lanthanides might lead to a formation of a different 3d-4f coordination complex under other crystallization conditions, which is for their size more stable.) Compounds 1-4 are stable under air and moisture. According to TGA curves (see Supporting Information), the solvent-free compounds 1-4 only degrade above ca. 220 °C under N2 atmosphere or in air, and in that they are slightly more stable than the HL·SMe ligand (ca. 200 °C). It is noteworthy that changing the reaction components in the above-mentioned synthetic process by replacing $Ln(NO_3)_3 \cdot 6H_2O$ with $Ln(OOCMe)_3 \cdot 4H_2O$ and $[Cu_2(OOCMe)_4(H_2O)_2]$ with $Cu(NO_3)_2\cdot 3H_2O$ does not result in compounds 1-4. Although these reactions are characterized by the same color gradient, they produce a neutral mononuclear compound 6 with the formula [Cu(L·SMe)₂] (for details see the Supporting Information), likely due to the lower solubility of the lanthanide acetate precursors (vs. the lanthanide nitrates). This complex can also be obtained by the direct reaction of the Schiff base ligand with common copper(II) salts in a 1:1 ratio in methanol.

The positive ion-mode ESI mass spectra of acetonitrile solutions of compounds 1–4 (see Supporting Information) exhibit the molecular mass peak of the $[CuLn(L\cdotSMe)_2(OOCMe)_2]^+$ fragment without a NO₃⁻ ion at m/z 814.012 (4) – 889.059 (3). These molecular masses are

detected in different intensities depending on the particular lanthanide ion (1 = 69 %; 2 = 100 %; 3 = 52 % and 4 = 100 %). Interestingly, the m/z patterns also indicate the presence of the $[CuNa(L\cdot SMe)_2]^+$ fragment at m/z 630.068 – 630.086 and of the $[Cu_2(L \cdot SMe)_3]^+$ fragment at m/z 944.081 – 944.110. The molecular mass peak of $[CuNa(L\cdot SMe)_2]^+$ demonstrates that complexes 1-4 are coordinatively labile against sodium salts, as also evident from the synthesis (Scheme 1). The detection of this fragment by ESI-MS and the square-pyramidal Cu^{II} environment with one vacant coordination side in 1-4 prompted us to tune the structural and physical properties of these complexes by reacting them with simple sodium salts such as NaOMe and NaN₃. The addition of these to a methanolic solution of freshly prepared 1-4 in a 2.6 : 1.0 ({CuLn} : ligand) molar ratio leads to the formation of a trinuclear compound with the formula $[Cu_2Na(L\cdot SMe)_4]NO_3\cdot 3MeOH$ (5), which in its solvent-free state is slightly less thermally stable (up to ca. 200 °C) than 1-4 (see Supporting Information). 5 can also be obtained directly by reacting HL·SMe with triethylamine and $Cu(NO_3)_2 \cdot 3H_2O$ in a 2.0 : 2.3 : 1.0 ratio at 65 °C in MeOH and the subsequent reaction of the formed dark-brown solution with 1.0 eq. of NaOMe under reflux conditions. The ESI-MS spectrum of an acetonitrile solution of compound 5 (see Supporting Information) shows the molecular mass peak of $[CuNa(L\cdot SMe)_2]^+$ at m/z 630.068 with 100 % intensity. Additionally, the mass spectrum displays the expected molecular mass peak of the monocationic fragment $[Cu_2Na(L\cdot SMe)_4]^+$ at m/z 1239.144 with 50 % intensity. Although m/z 1237.146 is expected to be the most intense monoisotopic mass, due to the isotopic distribution several signals around m/z 1239 add up to a more intense peak than the former.

X-ray Diffraction Structural Analysis. Since compounds 1-4 are quasi-isostructural (Figure 1) and crystallize in the triclinic space group P-1 (see Tables S2 and S3 in the Supporting Information), we here describe the structural parameters of only the Gd derivate (1) as a representative example. All neutral bimetallic complexes $[CuLn(L\cdot SMe)_2(OOCMe)_2(NO_3)]$ (Ln = Gd, Tb, Dy and Y) comprise a nine-coordinated lanthanide(III) or yttrium(III) ion and a copper(II) ion in a square-pyramidal N₂O₃ coordination environment. The metal centers are bridged by two deprotonated tridentate Schiff base ligands (L-SMe⁻) and an acetate ligand (Gd–O_{acetate}: 2.316(4) Å; Cu–O_{acetate}: 2.196(4) Å). The coordination polyhedron around the lanthanide (or yttrium) center is completed by chelating terminal nitrate (Gd– O_{NO3} : 2.508(4) – 2.517(4) Å) and acetate ligands (Gd– $O_{acetate}$: 2.432(4) – 2.476(4) Å). Each Schiff base ligand L SMe⁻ is attached to the Ln^{III}/Y^{III} center via its –OMe group (Gd– O_{ether} : 2.474(4) – 2.578(4) Å) and deprotonated O_{alc} atom of the alcohol group at the aryl ring (Gd– O_{alc} : 2.302(4) – 2.353(4) Å). The latter and an imine group of the L·SMe⁻ bind to the Cu^{II} ion (Cu– O_{alc} : 1.957(4) – 1.962(4) Å and Cu-N_{imine}: 1.998(5) - 2.012(4) Å). The non-bonding Gd…Cu distance is 3.3960(8) Å. The S atoms of two thioether groups at the periphery of the structure are separated by 5.57 Å. Importantly, these thioether groups are not involved in any intermolecular coordinative bond in the crystal lattice. $^{^{26,27}}$



Figure 1. Molecular structure of compounds **1–4**. Hydrogen atoms and crystal solvent molecules are omitted for clarity. Color code: C of L·SMe⁻: gray, C of acetate: green, Cu: brown, Ln/Y: turquoise, N: blue, O: red, S: yellow.

Compound 5 crystallizes in the monoclinic space group P_2/c (see Supporting Information, Table S3). The monocationic $[Cu_2Na(L\cdot SMe)_4]^+$ fragment of this complex shows a nearly linear structure with a Cu-Na-Cu angle of 178.75(6)° (Figure 2). The structure consists of two Cu^{ll}ions in distorted planar N₂O₂ environments separated by an octacoordinated sodium ion, with non-bonding Cu-Na and Cu-Cu distances of 3.407 Å and 6.813 Å, respectively. The molecular structure is supported by four L·SMe⁻ ligands, with two remote thioether groups lying roughly in the same plane as the metal centers. The shortest S···S distances are 5.04 Å and 6.25 Å, while the longest one between two sulfur atoms at opposite sides of $[Cu_2Na(L\cdot SMe)_4]^+$ is 19.46 Å. Each Cu^{II} center is coordinated by two nitrogen atoms (Cu-Nimine: 1.964(3) -1.972(4) Å) and two deprotonated O_{alc} atoms of the alcohol groups (Cu-O_{alc}: 1.887(3) - 1.903(3) Å) at the aryl rings of the adjacent L·SMe⁻ ligands. The distorted coordination environment of the central Na⁺ ion is saturated by four deprotonated O_{alc} atoms (Na– O_{alc} : 2.319(3) – 2.415(4) Å) and four -OMe groups (Na-O_{ether}: 2.568(4) - 2.685(4) Å) of all $L \cdot SMe^{-1}$ ligands. The charged $[Cu_2Na(L \cdot SMe)_4]^+$ species is counterbalanced by a NO_3^- anion.



Figure 2. Molecular structure of $[Cu_2Na(L-SMe)_4]^+$ in **5.** Hydrogen atoms, crystal solvent molecules and a nitrate counterion are

omitted for clarity. Color code: C = gray, Cu = brown, N = blue, Na = lime green, O = red, S = yellow.

Magnetism and Magnetochemical Modeling. The magnetic susceptibilities of compounds 1-5 are shown in Figure 3 as $\chi_m T$ vs. T and M_m vs. B plots. At 290 K and 0.1 T, the $\chi_m T$ values of the four compounds 1-4 are well within or close to the upper limit of the expected range for a copper center and the respective lanthanide center, which are not interacting: 1: 8.44 (expected:²⁸ 7.97 - 8.42), 2: 12.08 (12.0 -12.5), **3**: 14.14 (13.4 – 14.7), **4**: 0.46 (0.36 – 0.61) cm³ K mol⁻¹. Upon decreasing the temperature, the $\chi_m T$ curves of the compounds reveal different characteristics. For 4, where a diamagnetic Y³⁺ center substitutes the paramagnetic Ln³⁺ centers of 1–3, $\chi_m T$ gradually decreases to 0.43 cm³ K mol⁻¹ at 14.0 K, and subsequently rapidly decreases to $0.40 \text{ cm}^3 \text{ K mol}^{-1}$ at 2.0 K. While the first decrease is due to the single-ion effect of the Cu²⁺ center (thermal depopulation of the energy states split by a quadratic pyramidal ligand field in addition to mixing of these states due to spinorbit coupling), the second cannot be caused primarily by Zeeman splitting, considering the weak applied field of 0.1 T, but is most likely due to very weak inter-molecular exchange interactions present within the solid state. The molar magnetization at 2.0 K increases to 1.0 $N_A \mu_B$ at 5.0 T without reaching saturation. For 1, the Gd^{3+} center is, to a very good approximation, a pure S = 7/2 center. By cooling the compound, $\chi_m T$ continuously increases and shows three maxima, dependent on the applied field $(9.97 \text{ cm}^3 \text{ K mol}^{-1} \text{ at})$ 0.1 T and 5.5 K, $9.60 \text{ cm}^3 \text{ K} \text{ mol}^{-1}$ at 1.0 T and 10.0 K, $8.99 \text{ cm}^3 \text{ K mol}^{-1}$ at 3.0 T and 20.0 K), indicating ferromagnetic exchange interactions between the Cu²⁺ and the Gd³⁺ center. The shift of these maxima to higher temperatures with increasing fields, and the subsequent sharp decrease of $\chi_m T$ are due to the Zeeman splitting and the corresponding thermal depopulation of the energy states of both centers. As for 4, the molar magnetization of 1 is not saturated at 2.0 K and 5.0 T. The respective value of 8.0 $N_A \mu_B$ is, however, close to the expected saturation value of ca. 8.1 $N_A \mu_B$ $(M_{m,sat} = (g_{Cu} \cdot S_{Cu} + g_{Gd} \cdot S_{Gd}) N_A \mu_B \approx (1.1 + 7.0) N_A \mu_B), g_{Cu} \approx 2.2$ derived from the $\chi_m T$ value of **4** at 290 K). For **2**, $\chi_m T$ stays almost constant down to 100 K, slightly decreases upon further cooling to 10 K, and drops sharply below 10 K. We note the small change of the slope at about 30 K and the very sharp drop-off for T < 10 K, which hints at weak ferromagnetic exchange interactions between the copper and Tb^{3+} centers. This is because the $\chi_m T$ vs. T curves of single Tb³⁺ centers, characterized by similar ligand fields, exhibit a more distinct decrease starting notably at T < 50 K, and reach lower values at about 2.0 K due to the thermal depopulation of the (usually mixed) m_j substates. The molar magnetization at 2.0 K is linear up to ca. 0.5 T, and reaches a value of $6.3 N_A \mu_B$ at 5.0 T. At this point, a significant slope characterizes the magnetization, which is therefore far from saturation. We estimate the contribution of the Tb³⁺ center for the given coordination geometry at 5.0 T as approximately half of the saturation value of the free Tb³⁺ ion $(M_{m,sat} = g_J \cdot J N_A \mu_B = 9 N_A \mu_B)$ due to measuring the mean value (powder sample) of an anisotropic center. Taking into

account the magnetization of the latter and 1, the value of $M_{\rm m}$ at 5.0 T is slightly above the sum of both contributions. Therefore, the field dependent data at 2.0 K are also in agreement with no or weak ferromagnetic exchange interactions between the Cu^{2+} center and the Tb^{3+} center. For 3, $\chi_m T$ continuously decreases to a minimum at 20.0 K with decreasing temperature, subsequently increases to a maximum at 5.5 K, and finally drops off sharply. In this case, the ferromagnetic exchange interactions between the Cu²⁺ center and the Dy³⁺ center are evident from the occurrence of the distinct maximum. The M_m vs. B curve at 2.0 K is similar to the curve of 2 characterized by a steeper increase of the magnetization at lower fields. At 5.0 T, $M_{\rm m}$ is 6.4 $N_{\rm A} \mu_{\rm B}$, slightly larger than the sum of the magnetization of 1 and half of the saturation magnetization of the free Dy³⁺ ion $(M_{m,sat} = 10 N_A \mu_B)$. Thus, the magnetization data are in agreement with the weak ferromagnetic exchange interactions deduced from the $\chi_m T$ vs. T curve.



Figure 3. Temperature dependence of $\chi_m T$ (top) and field dependence of the molar magnetization M_m (bottom) of 1–4; open symbols: experimental data at 0.1 T (top) and 2.0 K (bottom), respectively; solid red lines: least-squares fits.

To quantify the underlying magnetically relevant properties, we model the data employing the computational framework CONDON,^{29,30} which takes into account interelectronic repulsion, ligand field, spin-orbit coupling, Zeeman effect and Heisenberg-Dirac-van Vleck exchange interactions, by implementing the following strategies. We start

by modeling the data of compound 4 to characterize the Cu^{2+} centers in 1–4. To generate starting values of the ligand field parameters, we assumed a ligand field symmetry of approximately C_{4v} during the calculations using the point charge electrostatic model (PCEM). While fitting the parameters to the data using the full basis of the 3d⁹ electron configuration (10 energy states), the relation B_4^4/B_0^4 was initially treated as constant. When the already good quality of the fit (SQ, relative root mean square error) did not improve any further, the relation was allowed to vary, yet only small deviations were found. During these steps, the signs of these parameters were set as derived from the PCEM. Finally, to account for the rapid decrease of $\chi_m T$ at T < 14.0 K, a mean-field approach was chosen to model potential weak inter-molecular exchange interactions. The parameters of the least-squares fit are listed in Table 1. The parameters describe a Cu²⁺ ion in a square pyramidal ligand field, which exhibits very weak antiferromagnetic, intermolecular exchange interactions (characterized by zJ'). For the analyses of 1-3, we assume the Cu²⁺ center to be identical to the one in **4**. We neglect, however, the very weak inter-molecular interactions, since the data here are dominated by the exchange interactions between the Cu²⁺ and Ln³⁺ centers.

Table 1. Magnetic quantities and fit parameters of **1–5**: oneelectron spin-orbit coupling constant ζ , Racah parameters *B* and *C*, Slater-Condon parameters F^2 , F^4 and F^6 , ligand field parameters B_q^k in Wybourne notation, mean-field (*zJ'*) and exchange interaction (*J*) parameters (both in "–2*J*" notation), all of which are stated in cm⁻¹.

	Cu ²⁺	Gd ³⁺	Tb ³⁺	Dy ³⁺	Cu ²⁺ (5)
	(1-4)	(1)	(2)	(3)	
$\zeta^{_{31,32}}$	829		1705	1900	829
B ³¹	1238				1238
C ³¹	4659				4659
F^{2} 32			97650	94500	
F ^{4 32}			68531	66320	
F ^{6 32}			52397	50707	
B ² o	20871 ± 15		-467 ± 3	-1207 ± 39	–17837 ± 2946
B ⁴ o	26579 ± 11		-233 ± 9	-1790 ± 72	13566 ± 1636
B ⁴ ₄	45905 ± 9		-294 ± 6	-1182 ± 11	-49941 ± 402
B ⁶ o			152 ± 4	126 ± 46	
B ⁶ ₄			2770 ± 4	1464 ± 63	
g _{eff}		1.99 ± 0.01			
zJ'	-0.09 ± 0.01				
J		+2.4 ± 0.6	+4.2 ± 1.5	+2.2 ± 1.0	-0.14 ± 0.09
SQ	1.2 %	0.7 %	0.5 %	0.3 %	0.3 %

In a next step, we model the data of compound 1 to estimate the strength and magnitude of the exchange interaction between the Cu^{2+} center and the Ln^{3+} centers of 1–3. Due to the well isolated orbital singlet ground state ⁸A₁ of Gd^{3+} centers, the Gd^{3+} center of 1 was modeled as an isotropic spin center with effective spin $S_{eff} = 7/2$ and g_{eff} slightly less than the g factor of the free electron due to mixing of excited states into the ground state. The found exchange interaction parameter of +2.4 cm⁻¹ indicates ferromagnetic exchange interactions between the Cu²⁺ center and the Gd³⁺ center, i.e. in the typical range for 3d-4f exchange interactions.³³ We employed the same strategy for the fitting procedure of the parameters of 2 and 3: Similar to 4, the starting values of the ligand field parameters were generated by the PCEM assuming a ligand field of approximately C_{4v} symmetry (in this case representing a capped square antiprism). The relations B_4^4/B_0^4 and B_4^6/B_0^6 were initially kept constant, and - after no further improvement of SQ – were allowed to vary while retaining the sign of the parameters. The starting value for the exchange coupling was set to the value of J as estimated for 1. While the full basis of a $4f^{N}$ electron configuration was used for the calculation of the single ion effects (2(N = 8): 3003 states and 3(N = 9): 2002 states, respectively), this basis was reduced to the 2J+1 states (2: 13, 3: 16) of the ground multiplet in addition to the 10 states of the Cu^{2+} center when considering the exchange interactions. The values of B_q^k and J of the corresponding least-squares fits are shown in Table 1. The ligand field parameters describe the Tb³⁺ center or the Dy³⁺ center, respectively, as a distorted capped square antiprism. The exchange interactions are ferromagnetic, and of same magnitude ($\sim 2-3 \text{ cm}^{-1}$) within the error margins.

The magnetic data of **5** are shown as $\chi_m T$ vs. *T* curve at 0.1 T and M_m vs. *B* curve at 2.0 K in Figure 4. The $\chi_m T$ value of 0.84 cm³ K mol⁻¹ at 290 K is within the expected range²⁸ of 0.72–1.21 cm³ K mol⁻¹ for two non–interacting Cu²⁺ centers. By decreasing temperature, $\chi_m T$ slightly decreases to 0.82 cm³ K mol⁻¹ at 18.0 K, and subsequently drops down to 0.78 cm³ K mol⁻¹ at 2.0 K. This drop is potentially due to very weak antiferromagnetic exchange interactions between the two Cu²⁺ centers. The molar magnetization at 2.0 K continuously grows by increasing the applied magnetic field *B*. At 5.0 T, M_m is 2.0 N_A μ_B and not saturated.



Figure 4. Temperature dependence of $\chi_m T$ at 0.1 T, and field dependence of the molar magnetization M_m at 2.0 K (inset) of 5; open symbols: experimental data; solid red lines: least-squares fits.

To model the data of 5 using CONDON, we assume both Cu²⁺ centers to be identical, in line with the molecular structure. The geometry of the ligand field of both centers is approximated as a tetragonal distorted tetrahedron (D_{2d}) . Starting values of the ligand field parameters were calculated by applying the PCEM, and all 10 states of the 3d⁹ electron configuration were considered per center during the fitting procedure. The result of the least-squares fit of quality SQ = 0.3% are given in Table 1. The two Cu²⁺ centers exhibit ligand field parameters that describe a tetragonal strongly distorted tetrahedral or almost quadratic planar coordination of the central ion. The exchange coupling constant is very small, representing an antiferromagnetic exchange interaction. The small magnitude of J is consistent with the large distance between both Cu²⁺ ions (6.813 Å).

FT-IR and FT-IRRAS Spectra. HL·SMe was first immobilized in the form of self-assembled monolayers (SAMs) on an Au surface from a 1.0 mmol ethanolic solution to assess the structural integrity of the uncoordinated, charge-neutral ligand on the solid substrate. As the comparison of the FT-IR and FT-IRRAS spectra of HL·SMe (see Supporting Information) indicates that its chemical structure remains effectively unchanged upon adsorption,³⁴ we subjected complexes 1-4 to IRRAS analysis. Although a small shift of the recorded IR vibrational bands is observed due to the presence of different Ln³⁺ ions in the respective compounds and thereby the resulting change in bond strengths, which is directly linked to the vibration frequency, the FT-IR spectra of 1-4 are nearly identical, as expected for these guasiisostructural complexes (see Supporting Information). Therefore, we here discuss as a typical example only the results of FT-IRRAS measurements of the Tb-containing sample (2). As can be seen in Figure 5, the similarity of the vibration frequencies of the recorded FT-IR and FT-IRRAS peaks in the fingerprint region, which arise mainly due to =C-H in- and out-of-plane deformation as well as C=C stretching vibrations of the aryl rings, suggests an intact immobilization of complex 2 on the Au surface. Due to the surprisingly high quality of the first-measured FT-IRRAS spectrum (denoted as "IRRAS 1" in Figure 5) and the small amount of solvent used for washing the Au substrate dropwise, we assume to have a thin layer of compound 2 on the gold surface. Subsequently, this Au substrate was dipped into methanol to wash the surface more carefully and a second FT-IRRAS spectrum ("IRRAS 2" in Figure 5) was recorded. The intensity of the obtained peaks decreases (as expected) and the remaining signals indicate that compound 2 still forms an intact thin layer on the substrate - presumably a monolayer. The insignificant difference in the wavenumbers of the peaks in the FT-IR and FT-IRRAS spectra of 2 (Table 2) is associated with the different sample forms (KBr vs. Au substrate) and the applied measurement methods (through-beam vs. reflective).³⁵



1195

1109-1080

IR

 \tilde{v} / cm⁻¹

3062

1612

1560

1489-1381

1463–1381

1381 and

1198-1184

1094 and

1298

1237

1078

Large-Area Charge Transport Measurements. In conjunction with the IRRAS results, we further investigated the propensity of compounds 1–4 to form SAMs on a gold substrate as bottom electrode and thereby of molecular tunneling junctions by employing eutectic Ga–In (EGaIn) as top electrode.¹⁰ EGaIn has proven to be instrumental in characterizing large-area junctions comprising a wide variety of SAMs. It is able to distinguish between details of the orientation of terminal methyl groups in alkanethiols,³⁷ resolve conformation-driven quantum interference in aromatic SAMs¹³ and help determine the orientation of SAMs of proteins.³⁸

We succeeded in growing SAMs of the target compounds by immersing freshly cleaved template-stripped gold substrates (atomically smooth Au^{TS})³⁹ in a ~ 0.1 mM methanolic solution of each metal complex overnight (see Supporting Information for details). After rinsing with pure methanol and drying in a gentle stream of nitrogen, the SAMs were contacted with EGaIn tips to form junctions of the structure Au^{TS} //SAM//Ga₂O₃/EGaIn where "/" denotes interface defined by chemisorption and "//" by physisorption. Note that



Figure 5. IRRAS 1 (red), IRRAS 2 (black) and IR (blue) spectra of compound 2 in the $1800-1000 \text{ cm}^{-1}$ region.

Table 2. A comparison of selected band vibrations between bulk IR (as KBr pellet) and IRRAS (on gold substrate) spectra of compound $2.^{36}$

IRRAS 2

 \tilde{v} / cm⁻¹

1553-1547

1463-1442

1391 and 1382

3007

1612

1239

1192

1106

assignment

 $\tilde{V}_{ar}(C-H)$

 $\tilde{V}_{ar}(C=C)$

 $\tilde{V}_{as}(COO^{-})$

 $\tilde{V}_{s}(COO^{-})$

 $\tilde{V}_{ar}(C=C)$

 $\tilde{V}_{\varsigma}(COO^{-})$

 $\delta_{iD}(=C-H)$

 $\delta_{iD}(=C-H)$

 $\delta_{ip}(=C-H)$

EGaln is an eutectic alloy of Ga and In (75.5% Ga and 24.5% In by weight), the surface of which is covered by thin, conductive, self-limiting layer of Ga_2O_3 .⁴⁰ Along with EGaIn we used ellipsometry and STM measurements to characterize the SAMs (see Supporting Information for details). Although we observe the formation of a monolayer of complexes, this type of self-assembly should not be confused with the densely-packed, upright SAMs of thiolates that form from, for example, alkanethiols. The characterization that we provide (e.g., ellipsometry and FT-IRRAS) only shows that the complexes are immobilized on the surface in a disordered monolayer. They lay flat, not upright, and are rotated randomly about the surface-normal axis and the identity of the complex does not substantially alter the structure of the monolayer.

Finally, we performed an analysis of the current-voltage (1-V) characteristics of the engineered SAMs. The results of our I-V measurements are illustrated in Figure 6. Apart from minor differences in the shape of J-V curves (current density J = I/S, where S is the area of the junction) the four SAMs are indistinguishable. By replotting J-V data in Fowler-Nordheim coordinates (transition voltage spectroscopy⁴¹) it is possible to obtain information about energy level alignment inside the junction. All values of transition voltages (V_T) coalesce to ~0.3 V (see Supporting Information). This value can be attributed to the $\beta d_{x-y^2}^{2}$ main transmission channel of Cu, which lies close to the Fermi level and is shared by all compounds 1-4 (see DFT section below for details). A similar value of V_T was previously ascribed to Ga₂O₃ in junctions comprising SAMs of alkanethiolates;⁴² however, the frontier orbitals of the alkane backbone are much higher in energy than Ga_2O_3 , which is not the case for the complexes in this study. Moreover, the transition voltages for SAMs with accessible frontier orbitals have been unambiguously assigned in EGaIn junctions⁴³ and the value of ~0.3 V is likely a numerical coincidence. Assigning the transition voltage to the βd_{x-y}^{2-2} main transmission channel of Cu is also consistent with a single-level model that was proposed for ferrocene-containing molecules in which the Fe center mediates transport.⁴⁴



Figure 6. Plots of the logarithmic current density versus applied potential for SAMs of the compounds 1-4. Values of $\log|J|$ at V = 0 V are omitted for clarity. Error bars represent the standard devia-

7

tion of Gaussian fits. Four traces are indistinguishable at full bias range.

Most SAMs studied previously by EGaIn are chemically bound to the substrate (usually by sulfur-metal bonds). However, in compounds 1-4 the sulfur atoms are divalent (thioether) and can only weakly interact with the metal surface through physisorption. $^{\rm 45-47}$ Thus we expect i) the SAM to be poorly ordered and ii) the molecules to be weakly coupled to the bottom electrode, that is to exhibit high resistance. Due to the absence of a free thiol group to bind to the bottom substrate it was not clear whether these complexes will form a SAM and if so, whether it is possible to measure them in large-area junctions. However, all compounds 1-4 formed surprisingly electrically robust monolayers with the average yield of working EGaIn junctions of 67%. This observation may seem counter-intuitive; however, there is evidence that disordered, liquid-like SAMs yield better data, because molecular motion is much faster than the time scale of the measurement.⁴⁸ We are not suggesting that the complexes are liquid-like, only that the degree of order in a SAM is not positively correlated to the quality of the data obtained from large-area junctions measured using EGaIn.



Fig. XX β -plot for the series of alkanethiolates on Ag^{TS} with the J_{avg} of compounds 1–4. Length for alkanethiolates is the Sulphur-Hydrogen distance measured from MM2 minimized structure. Length *a* corresponds to the shortest dimension of the complexes measured from the crystal structures and length *b* corresponds to the longest, length *c* corresponds to the average ellipsometric thickness.

To assess the conductivity of compounds 1–4 we determined the average conductance of the combined data and compared it to the benchmark system for the SAM-based large-area molecular junctions – alkanethiolates on Ag (see Supporting Information). It was previously shown that SAMs of even-numbered alkanethiols on Ag^{TS} and Au^{TS} exhibit identical transport properties in EGaIn junctions,⁴⁹ which allows the comparison of the data across the substrates. To estimate the thickness of the monolayers we performed ellipsometric measurements on the monolayers and used crystal structures of the complexes. The data are consistent with the thickness of ~1.3 nm. As expected, due to the higher contact resistance at the bottom interface, the average conductivity of our SAMs is lower (~2-3 orders of magnitude) than that for alkanethiolates of equivalent length.

DFT Study of Transport Properties. To obtain a better understanding of the collected current–voltage data showing the similar conductivity behavior for all studied SAMs, we performed quantum mechanical calculations. However, the DFT combined with equilibrium Green function (EGF) was employed to calculate the coherent transport properties of only complexes 1 and 4. The reason for this is the single-determinant nature of the DFT, which limits the description of the f-type ions because of their degenerated ground state.⁵⁰ Although some information about the electronic structure of the f-type ions can indeed be extracted from DFT calculations, we have restricted our calculations to the non-degenerated ground-state ions Gd^{III} and Y^{III}.

To the best of our knowledge, no X-ray structure to build up the EGaIn electrode is available. Nonetheless, for this kind of Metal–SAM–EGaIn junctions it has been shown that the SAM and not the electrodes dominates the charge transport.⁴⁰ Due to this fact, the molecular structure of **1** and **4** as determined by single-crystal X-ray diffraction was embedded between two Au(111) electrodes as shown in Figure 7, thus simulating the experimental two-terminal setup. As the interaction between neighboring molecules is weak according to our experimental data, a single molecule approach is hence well suited to calculate the transmission spectra.



Figure 7. A theoretical setup to examine transport properties of compounds 1 and 4. The explicit asymmetry in the gold electrodes definition was introduced to avoid overlapping between neighboring atoms. Color code: C of L-SMe⁻: gray, C of acetate: green, Cu: brown, Gd/Y: turquoise, N: blue, O: red, S: yellow.

The electronic structure calculation of 1 suggests a ferromagnetic ground state interaction between the Cu and Gd magnetic centers, in agreement with the experimental data. The transmission spectra (T(E)) of 1 and 4 and their projected density of states (PDOS) on the molecule are shown in Figures 8 and 9, respectively.



Figure 8. PDOS on the molecule (left) and transmission spectrum log(T(E)) of 1 (right). The Fermi energy is set to zero. Red and blue colors stand for α and β spin orbital contributions. The shaded peaks correspond to the Cu atom contribution to the molecular orbital. Gd has almost no contribution around the Fermi level.

The PDOS of complex 1 (Fig. 8, left) shows a single peak at 0.1 eV above the Fermi level $(E-E_F = 0)$ that corresponds to a molecular level with a high contribution of the β d_x2_{-v}2 atomic orbital of the Cu center. As can be seen in the transmission spectrum (Fig. 8, right), this molecular spin orbital has its corresponding transmission peak. Because of its proximity to the Fermi level, this energy level constitutes the main transmission channel in the junction. Complex 4 presents a very similar behavior as illustrated in Figure 9. We note again that Cu atom has a strong contribution to the transmission near the Fermi level as in the {CuGd} analogue and the distance to the Fermi level is the same. Moreover, both the PDOS and the transmission are almost identical for both cases for the shown energy range. Thus, the transmission spectra are to a large extent fully independent on the lanthanide atom for a large energy range. This is in excellent agreement with the indistinguishable I-V curves obtained for complexes 1-3 (Ln = Gd, Tb, Dy) and 4 (Ln = Y). Because of the β -polarized transmission, we expect to observe spin-filtering properties for all set of the studied metal complexes upon their future contact with a magnetic electrode.



Figure 9. PDOS on the molecule (left) and transmission spectrum log(T(E)) of **4** (right). The Fermi energy is set to zero. Red and blue colors represent α and β spin orbital contributions. The shaded

peaks correspond to the Cu atom contribution to the molecular orbital. Y has almost no contribution around the Fermi level.

CONCLUSIONS

We have demonstrated that the charge-neutral 3d-4f complexes in 1-4, exhibiting air, moisture and thermal stability, serve as a suitable materials platform for the study of largearea molecular transport properties. Although we cannot unequivocally rule out the possibility of multilayer formation based on our IRRAS experiments, the acquired STM and ellipsometry data point to densely packed, disordered monolayers of the {Cu^{II}Ln^{III}} title complexes. Remarkably, varying the lanthanide ion from Gd^{3+} (1) to Tb^{3+} (2) and Dy^{3+} (3) does not have a measurable effect on the conductivity of these SAMs in Au^{TS}//SAM//Ga₂O₃/EGaIn junctions, most likely due to the fact that their 4f states are too deep in energy. We conjecture that only early lanthanides, with their 4f states close to the Fermi edge, would be able to significantly affect the molecular charge transport.⁵¹ The structural constraints of the present ligand system, however, preclude the integration of such larger early lanthanide ions. Our DFT+EGF calculations indicate that the tunneling transport should occur through the molecular spin orbital of the copper center forming the main transmission channel across the molecular junction. The results obtained open up far-reaching opportunities for investigation of this type of molecular structures toward their binuclear 3d-4f congeners by changing Cu for another 3d-metal spin center. Such a modification of the transition metal ion may have substantial effects on the electronic and magnetic picture of the molecule, thus influencing conductivity and potential spin-filtering behavior when applying a magnetic electrode.

EXPERIMENTAL AND COMPUTATIONAL METHODS

Materials and methods. The syntheses of the Schiff base ligand HL-SMe and compounds **1–6** were carried out under aerobic conditions. All commercial starting materials were used as received. Solvents were used without further purification. CHN analysis was performed using a Vario EL elemental analyzer. IR spectra of HL-SMe and **1–6** were recorded on a Nicolet Avatar 360 FTIR spectrometer (KBr pellets, $m_{\rm KBr} \approx 250$ mg) in the range $\tilde{V} = 4000-400$ cm⁻¹. TGA curves for HL-SMe and **1–5** were obtained in air and under a nitrogen atmosphere with a heating rate of 5 K min⁻¹ in the temperature range 25–800 °C by using a Mettler Toledo TGA/SDTA 851e instrument. The ESI-MS spectra of HL-SMe and **1–6** in the positive ion mode were recorded on a 4000 QTRAP mass spectrometer system by using the LC/LC-MS method with direct infusion.

Synthesis of the Schiff base (HL-SMe). 2-Hydroxy-3methoxybenzaldehyde (*ortho*-Vanillin) (3.738 g, 24.6 mmol) was dissolved in 100 mL of ethanol. 4-(methylthio)aniline (3.0 mL, 24.6 mmol) was added to the yellow solution, resulting in a color change to orange. The solution was acidified with 5 drops of acetic acid, to catalyze the reaction. After stirring under refluxing conditions for 5 hours the ethanolic solution was cooled down to room temperature and stored in a flask under ambient conditions. Orange

needle-like crystals of HL·SMe were isolated after one day and washed with ice cooled ethanol and pentane. Yield of the air-dried crystals: 6.253 g (87.2 %). Elemental analysis, calcd. for $C_{15}H_{15}NO_2S \cdot 0.1$ pentane (M = 273.35 g·mol⁻¹ without crystal solvent): C, 66.35; H, 5.82 and N, 4.99 %. Found: C, 66.61; H, 5.66 and N, 5.15 %. **IR** (KBr pellet), \tilde{V}_{max} / cm⁻¹: 3442 (m, br), 3000 (vw), 2955 (w), 2921 (w), 2832 (vw), 1883 (vw), 1745 (vw), 1609 (s), 1571 (m), 1561 (sh), 1466 (s), 1439 (m), 1405 (sh), 1361 (m), 1326 (sh), 1271 (sh), 1257 (s), 1199 (m), 1180 (w), 1123 (w), 1091 (m), 1077 (m), 1008 (w), 968 (s), 936 (m), 861 (m), 833 (w), 821 (s), 812 (sh), 778 (m), 732 (s), 708 (sh), 679 (w), 582 (w), 547 (w), 503 (m), 421 (w). MS (MeOH, ESI): *m*/*z* = 296.071 (NaC₁₅H₁₅NO₂S⁺, 100 %), 274.089 $(HC_{15}H_{15}NO_{2}S^{\dagger}, 40\%)$. ¹H-NMR (in CD₂Cl₂, 400 MHz): $\delta = 13.43$ (s, 1H, OH), 8.66 (s, 1H, N=CH-ar), 7.33-7.26 (m, 4H, H-ar), 7.04 (dd, 1H, H–ar), 7.00 (dd, 1H, H–ar), 6.89 (t, 1H, H–ar), 3.90 (s, 3H, H₃C–O–R) and 2.51 (s, 3H, H₃C–S–R). ¹³C-NMR (in CD_2Cl_2 , 100 MHz): δ = 162.6, 151.9, 149.0, 145.8, 138.2, 127.8, 124.3, 122.2, 119.8, 119.0, 115.5, 56.7 and 16.3 ppm.

Synthesis of [CuGd(L·SMe)2(OOCMe)2(NO3)]·MeOH (1). The Schiff base HL·SMe (0.137 g, 0.5 mmol) was dissolved in 10 mL of methanol, and triethylamine (0.08 mL, 0.58 mmol) was introduced into the solution. $Gd(NO_3)_3 \cdot 6H_2O$ (0.113 g, 0.25 mmol) was added and the reaction mixture was refluxed for 15 minutes to give a clear orange solution. $[Cu_2(OOCMe)_4(H_2O)_2]$ (0.050 g, 0.13 mmol) was then added, which resulted in a color change to dark brown. The methanolic solution was refluxed for 1 hour, filtered off and the filtrate was stored in a capped vial at room temperature. Dark brown single crystals of compound 1 were isolated after one day and washed with a small amount of ice cold methanol. Yield of the air-dried crystals: 0.074 g (31.3 %). Elemental analysis, calcd. for $C_{34}Cu_1H_{34}Gd_1N_3O_{11}S_2$ (M = 945.57 g·mol⁻¹ without crystal solvent): C, 43.19; H, 3.62 and N, 4.44 %. Found: C, 42.88; H, 3.50 and N, 4.59 %. IR (KBr pellet), \tilde{v}_{max} / cm⁻¹: 3442 (m, br), 3062 (w), 2984 (w), 2919 (w), 2842 (w), 1612 (vs), 1560 (s), 1465 (vs), 1411 (s), 1381 (m), 1340 (w), 1298 (s), 1237 (s), 1198 (s), 1184 (m), 1094 (m), 1078 (m), 1048 (w), 1032 (w), 1013 (w), 969 (m), 931 (w), 853 (m), 824 (m), 789 (w), 735 (m), 712 (w), 678 (m), 652 (w), 612 (w), 584 (m), 540 (w), 446 (w). **MS** (MeCN, ESI): m/z = 630.068 ($C_{30}Cu_1H_{28}Na_1N_2O_4S_2^+$, 15 %; [$CuNa(L\cdot SMe)_2$]⁺), 883.031 $(C_{34}Cu_1H_{34}Gd_1N_2O_8S_2^+,$ 69 %; $[CuGd(L \cdot SMe)_2(OOCMe)_2]^+)$, 944.083 $(C_{45}Cu_2H_{42}N_3O_6S_3^+, 14)$ %; [Cu₂(L·SMe)₃]⁺), 1096.093 (C₄₇Cu₁H₄₅Gd₁N₃O₈S₃⁺, 100 %; $[CuGd(L\cdot SMe)_3(OOCMe)]^+).$

Synthesis of [**CuTb**(**L**-**SMe**)₂(**OOCMe**)₂(**NO**₃)]-**MeOH** (2). Compound **2** was synthesized following the procedure described for compound **1**, replacing Gd(NO₃)₃·6H₂O by Tb(NO₃)₃·6H₂O (0.114 g, 0.25 mmol). Yield of the air-dried crystals: 0.161 g (68.0 %). **Elemental analysis**, calcd. for C₃₄Cu₁H₃₄Tb₁N₃O₁₁S₂·H₂O (M = 947.25 g·mol⁻¹ without crystal solvent): C, 42.31; H, 3.76 and N, 4.35 %. Found: C, 42.33; H, 3.53 and N, 4.28 %. **IR** (KBr pellet), \tilde{v}_{max} / cm⁻¹: 3442 (m, br), 3062 (w), 2983 (w), 2919 (w), 2842 (w), 1612 (vs), 1560 (s), 1489 (s), 1464 (vs), 1411 (s), 1381 (m), 1298 (s), 1237 (s), 1198 (s), 1184 (m), 1094 (m), 1078 (m), 734 (m), 711 (w), 679 (m), 652 (w), 612 (w), 583 (m), 539 (w), 447 (w). **MS** (MeCN, ESI):

Synthesis of [CuDy(L·SMe)₂(OOCMe)₂(NO₃)]·0.75MeOH (3). Compound 3 was synthesized following the procedure described for compound 1, replacing $Gd(NO_3)_3 \cdot 6H_2O$ by $Dy(NO_3)_3 \cdot 6H_2O$ (0.115 g, 0.25 mmol). Yield of the air-dried crystals: 0.176 g (74.0 %). Elemental analysis, calcd. for $C_{34}Cu_1H_{34}Dy_1N_3O_{11}S_2 \cdot 0.75H_2O$ (M = 950.82 g·mol⁻¹ without crystal solvent): C, 42.35; H, 3.71 and N, 4.36 %. Found: C, 42.39; H, 3.64 and N, 4.38 %. **IR** (KBr pellet), $\tilde{\nu}_{max}$ / cm⁻¹: 3424 (m, br), 3063 (w), 2982 (w), 2918 (w), 2841 (w), 1614 (s), 1560 (s), 1490 (sh), 1463 (s), 1449 (sh), 1414 (sh), 1341 (w), 1305 (s), 1238 (s), 1199 (s), 1094 (m), 1078 (m), 1048 (w), 1033 (w), 1013 (w), 971 (m), 932 (w), 853 (m), 824 (m), 789 (w), 741 (m), 738 (sh), 712 (w), 680 (m), 653 (w), 612 (w), 585 (m), 539 (w), 449 (w), 405 (w). MS (MeCN, ESI): m/z = 630.068 ($C_{30}Cu_1H_{28}Na_1N_2O_4S_2^+$, 54 %; [$CuNa(L\cdot SMe)_2$]⁺), 889.059 $(C_{34}Cu_1H_{34}Dy_1N_2O_8S_2^+,$ 52 %: $[CuDy(L\cdot SMe)_2(OOCMe)_2]^+)$, 944.109 $(C_{45}Cu_2H_{42}N_3O_6S_3^+, 100)$ %; [Cu₂(L·SMe)₃]⁺), 1102.095 (C₄₇Cu₁H₄₅Dy₁N₃O₈S₃⁺, 56 %; $[CuDy(L \cdot SMe)_3(OOCMe)]^+).$

Synthesis of $[CuY(L\cdot SMe)_2(OOCMe)_2(NO_3)]$ ·MeOH (4). Compound **4** was synthesized following the procedure described for compound 1, replacing Gd(NO₃)₃·6H₂O by $Y(NO_3)_3 \cdot 6H_2O$ (0.096 g, 0.25 mmol). Yield of the air-dried crystals: 0.053 g (24.2 %). Elemental analysis, calcd. for $C_{34}Cu_1H_{34}Y_1N_3O_{11}S_2 \cdot 0.5H_2O$ (M = 877.23 g·mol⁻¹ without crystal solvent): C, 46.55; H, 3.91 and N, 4.79 %. Found: C, 46.13; H, 3.73 and N, 5.00 %. **IR** (KBr pellet), \tilde{v}_{max} / cm⁻¹: 3432 (w, br), 3062 (w), 2981 (w), 2921 (w), 2849 (w), 1613 (vs), 1560 (s), 1490 (s), 1474 (vs), 1414 (m), 1383 (m), 1341 (vw), 1305 (s), 1238 (s), 1198 (s), 1095 (m), 1074 (m), 1048 (vw), 1034 (w), 1015 (w), 971 (m), 933 (vw), 853 (w), 824 (w), 788 (vw), 745 (m), 712 (w), 681 (w), 652 (w), 614 (vw), 584 (w), 540 (w), 448 (w). **MS** (MeCN, ESI): *m*/*z* = 630.068 $(C_{30}Cu_1H_{28}Na_1N_2O_4S_2^+, 20\% [CuNa(L·SMe)_2]^+), 814.012$ $(C_{34}Cu_1H_{34}Y_1N_2O_8S_2^+, 100 \%; [CuY(L\cdot SMe)_2(OOCMe)_2]^+),$ 944.083 (C₄₅Cu₂H₄₂N₃O₆S₃⁺, 100 %; [Cu₂(L·SMe)₃]⁺), 1027.073 $(C_{47}Cu_1H_{45}Y_1N_3O_8S_3^+, 64\%; [CuY(L\cdot SMe)_3(OOCMe)]^+).$

Synthesis of [Cu₂Na(L·SMe)₄]NO₃·3MeOH (5). Method A: Compound 1 (0.095 g, 0.1 mmol) was dissolved in 10 mL of methanol under refluxing conditions and NaOMe (0.014 g, 0.26 mmol; alternatively: NaN₃ 0.017 g, 0.26 mmol) was added to the brownish solution. The solution was stirred at 65 °C for 1 hour. The methanolic solution was filtered off and the filtrate was stored in a capped vial at room temperature. Dark brown needle-like single crystals of compound 5 were isolated after one day, washed with a small amount of ice-cold methanol and dried in air. Yield: 0.023 g (34.9 %, based on Cu, no solvent). Method B: The Schiff base HL·SMe (0.137 g, 0.5 mmol) was dissolved in 10 mL of methanol and triethylamine (0.08 mL, 0.58 mmol) was introduced into the solution. $Cu(NO_3)_2 \cdot 3H_2O(0.061 \text{ g}, 0.25 \text{ mmol})$ was added, which gave a dark brown color. After stirring at 65 °C for 15 minutes NaOMe (0.014g, 0.26 mmol) was added, and the methanolic solution was then stirred under

refluxing conditions for further 30 minutes. The solution was filtered off and the filtrate was stored in a capped vial at room temperature. Dark brown needle-like single crystals of compound 5 were isolated after one day, washed with a small amount of ice cold methanol and dried in air. Yield: 0.033 g (20.0 %, based on Cu). Elemental analysis, calcd. for $C_{60}Cu_2H_{56}N_5Na_1O_{11}S_4 \cdot H_2O$ (M = 1319.47 g·mol⁻¹ disregarding solvent): C, 54.62; H, 4.43 and N, 5.31 %. Found: C, 54.23; H, 4.51 and N, 5.15 %. **IR** (KBr pellet), \tilde{v}_{max} /cm⁻¹: 3424 (m, br), 2918 (w), 2829 (w), 1606 (vs), 1581 (sh), 1542 (s), 1408 (s), 1466 (s), 1434 (s), 1382 (m), 1325 (s), 1236 (vs), 1191 (vs), 1107 (m), 1091 (m), 1076 (m), 1011 (w), 981 (m), 853 (m), 821 (m), 740 (m), 710 (sh), 683 (w), 578 (m), 539 (w), 441(w). MS (MeCN, ESI): m/z = 630.067 $(C_{30}Cu_1H_{28}N_2Na_1O_4S_2^+, 100\%; [CuNa(L \cdot SMe)_2]^+), 944.083$ $[Cu_2(L \cdot SMe)_3]^+),$ 1239.144

X-ray crystallography. Single-crystal diffraction data were collected on a Bruker APEX II CCD diffractometer at 100 K for 1 and on a SuperNova (Agilent Technologies) diffractometer at 120 K for **2–6** and HL·SMe with MoK α radiation $(\lambda = 0.71073 \text{ Å})$ for all the compounds except **4**, for which CuK α radiation (λ = 1.54184 Å) has been used. The crystals were mounted in a Hampton cryoloop with Paratone-N oil to prevent water loss. Absorption corrections for 1 were applied empirically using the SADABS program.⁵² Absorption corrections for 2–6 and HL·SMe were done numerically based on multifaceted crystal model using CrysAlis software.⁵³ The structures were solved by direct methods and refined by full-matrix least-squares method against $|F|^2$ with anisotropic thermal parameters for all non-hydrogen atoms (Gd, Dy, Tb, Y, Cu, S, O, N and C) employing the SHELXTL software package.⁵⁴ ISOR restrictions had to be applied for some carbons. Hydrogen atoms of the complexes and the ligand HL·SMe were placed in geometrically calculated positions, while the hydrogen atoms of the disordered solvent CH_3OH molecules (in 1–5) were not located.

The relative site occupancy factors for the disordered positions of carbon and oxygen atoms of co-crystallized methanol molecules in **1–5** were first refined in an isotropic approximation with U_{iso} = 0.05 and then fixed at the obtained values and refined without the thermal parameters restrictions. The relative occupancies of the disordered – C₆H₄–S–CH₃ moieties in **3** were refined using a combination of PART and EADP (for the heaviest S atoms) commands.

The relatively high residual electron density in the structure of **3** (2.404 eÅ⁻³) is located in the proximity of the Dy1 center (0.87 Å). The surprisingly high residual electron density maximum (3.61 eÅ⁻³) in the structure **6** cannot be reasonably assigned to any atom (*e. g.* solvent oxygen) due to its very small distance to carbon atoms of the complexes, C204 (1.538 Å) and C205 (1.789 Å). This apparently corresponds to some residual absorption artefacts.

Additional crystallographic data are summarized in Tables S2 and S3. Further details on the crystal structures investigation can be obtained, free of charge, on application to CCDC, 12 Union Road, Cambridge CB2 1EZ, UK: http://www.ccdc.cam.ac.uk/, e-mail: da-

ta_request@ccdc.cam.ac.uk, or fax: +441223 336033 upon quoting 1530688 (1), 1530689 (2), 1530690 (3), 1530691 (4), 1530692 (5), 1530693 (6) and 1530694 (HL·SMe) numbers.

Magnetic susceptibility measurements. Magnetic susceptibility data of compounds 1–5 were recorded using a Quantum Design MPMS-5XL SQUID magnetometer for direct current (dc) and alternating current (ac) measurements. The polycrystalline samples were compacted and immobilized into cylindrical PTFE capsules. The dc susceptibility data were acquired as a function of the field (0.1–5.0 T) and temperature (2.0–290 K). The ac susceptibility data were measured in the absence of a static bias field in the frequency range 10–1000 Hz (T = 2.0-50 K, $B_{ac} = 3$ G), but no out-of-phase signals were observed. The data were corrected for diamagnetic contributions from the sample holder and the compounds ($\chi_{m,dia} / 10^{-4}$ cm³ mol⁻¹, 1: –4.76, 2: –4.82, 3: –4.91, 4: –4.07, 5: –4.83).

IRRAS measurements. IRRAS measurements were performed on a FT-IR spectroscope Vertex 70, Bruker Optics equipped with a high-sensitivity Hg-Cd-Te (MCT) detector and an A513/Q variable angle reflection accessory including an automatic rotational holder for MIR polarizer. The IR beam was polarized with a KRS-5 polarizer with 99 % degree of polarization. Double-sided interferograms were collected with a sample frequency of 20 kHz, an aperture of 1.5 mm and a nominal spectral resolution of 4 cm⁻¹. The interferograms were apodized by a Blackmann-Harris 3-term apodization and zero-filled with a zerofilling factor of 2. The angle of incidence was set to 80°, and p-polarized IR radiation was used to record the spectra. For the background measurements, the sample chamber was purged with argon for 5 min, then 1024 scans were collected while continuing to purge. For the sample measurements, argon purging was started at the moment the first scan was recorded. The scans were averaged until the peaks arising from the water vapor in the sample chamber were compensated, for what typically 800–1500 scans were necessary. The spectra were processed using the OPUS software (Bruker). Where necessary, scatter correction was applied to the spectra.

General procedure for the preparation of Au substrates for IRRAS. The gold substrates were fabricated by sputtering a 10 nm adhesive film of Ti and a 100 nm thick layer of Au on <100> oriented silicon wafers with a native SiO₂ layer. The freshly prepared gold substrates were cleaned in oxygen plasma [$p(O_2) = 0.4$ mbar, f = 40 kHz and P = 75 W] for 4 min immediately prior to the deposition of molecules. The compounds studied by IRRAS were prepared for the deposition as follows: a solution (~ 1.0 mmolar) of HL-SMe was prepared using absolute ethanol and a solution of compound **2** using methanol GPR Rectapur (purity: 100 %). The Au substrates were stored for 24 h in the solutions and dried for 24 h in a desiccator.

DFT calculations. Transport properties of complexes 1 and 4 were studied using a combination of DFT and EGF. The mean-field Hamiltonian of both complexes was constructed using the SIESTA (Spanish Initiative for Electronic Simulations with Thousands of Atoms) code.^{55,56} The generalized-gradient approximation (GGA) functional expression of

Perdew, Burke and Ernzerhof (PBE)⁵⁷ was employed and valence pseudopotentials were generated according to the method suggested by Troullier and Martins⁵⁸ except for gold, where 1-electron pseudopotential was employed instead. Note that this pseudopotential gives incorrect structures if it is used for geometry optimization but reasonable transport properties in single-point calculations. A double-zeta basis set with polarization functions was used for all elements. Gollum code was exploited to perform the post-processing transport calculations.⁵⁹ The DFT-Hamiltonian and overlap matrices were mapped into a tight-binding scheme to compute the coherent transport properties. The EGF is the non-self-consistent method that presents a good compromise between accuracy and computational cost.60

ASSOCIATED CONTENT

Supporting Information. Synthesis and characterization of compound **6**; analytical data of HL·SMe; IR spectra of compounds **1–5** and IRRAS spectra of compound **2**; crystal data and structure refinement details for compounds **1–6** and HL·SMe; a comparative analysis of structural data between compounds **1–5** and related complexes described in the literature; measured ESI-MS spectra of compounds **1–5**; calculated isotopic pattern of compounds **1–5**; TGA curves of compounds **1–5**; details of large-area transport measurements. This material is available free of charge via the Internet at http://pubs.acs.org.

AUTHOR INFORMATION

Corresponding Author

* E-mails: kirill.monakhov@ac.rwth-aachen.de, paul.koegerler@ac.rwth-aachen.de, r.c.chiechi@rug.nl, eliseo.ruiz@qi.ub.edu.

Notes

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Table of Contents synopsis



The molecular conductivity of charge-neutral dinuclear 3d-4f coordination complexes that self-assemble into robust monolayers on gold electrodes highlights the distinct role of the metal ions in their molecular charge transport characteristics.