

1 **Chitosan-Derived Nitrogen-Doped Carbon Electrocatalyst**
2 **for a Sustainable Upgrade of Oxygen Reduction to Hydrogen**
3 **Peroxide in UV-Assisted Electro-Fenton Water Treatment**

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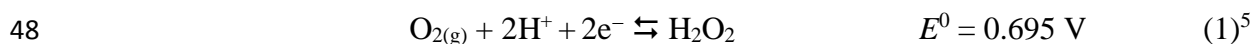
19 **ABSTRACT**

20 The urgency to move from critical raw materials to highly available and renewable feedstock is
21 currently driving the scientific and technical development. Within this context, the abundance of
22 natural resources like chitosan paves the way to synthesize biomass-derived nitrogen-doped
23 carbons. This work describes the synthesis of chitosan-derived N-doped mesoporous carbon in the
24 absence (MC-C) and presence (N-MC-C) of 1,10-phenanthroline, which acted as both porogen
25 agent and as second nitrogen source. The as-prepared MC-C and N-MC-C were thoroughly
26 characterized and further employed as catalytic materials in gas-diffusion electrodes (GDEs),
27 aiming to develop a sustainable alternative to conventional GDEs for H₂O₂ electrogeneration and
28 the photoelectro-Fenton (PEF) treatment of a drug pollutant. N-MC-C presented a higher content
29 of key surface N-functionalities like pyrrole group, as well as an increased graphitization degree
30 and surface area (63 vs. 6 m²/g), being comparable to commercial carbon black. These properties
31 entailed a superior activity of N-MC-C for the oxygen reduction reaction, as confirmed from its
32 voltammetric behavior at a rotating ring-disk electrode. The GDE prepared with N-MC-C catalyst
33 showed greater H₂O₂ accumulation, attaining values close to those obtained with a commercial
34 GDE. N-MC-C- and MC-C-derived GDEs were employed to treat drug solutions at pH 3.0 by PEF
35 process, which outperformed electro-oxidation (EO). The fastest drug removal was achieved using
36 N-MC-C, needing only 16 min at 30 mA/cm² instead of 20 min required with MC-C. The
37 replacement of the dimensionally stable anode by a boron-doped diamond (BDD) accelerated the
38 degradation process, reaching an almost complete mineralization in 360 min. The main
39 degradation products were identified, revealing the formation of six different aromatic
40 intermediates, alongside five aliphatic compounds that comprised three nitrogenated structures.
41 The initial N was preferentially converted into ammonium.

42 *Keywords:* Biomass; Electrochemical water treatment; Gas-diffusion electrode; Nitrogen-doped
43 carbon; Organic pollutant; Oxygen reduction reaction

44 INTRODUCTION

45 Lately, selected single atoms¹ as well as polyatomic Co-based compounds²⁻⁴ have been
46 successfully tested to electrocatalyze the two-electron oxygen reduction reaction (ORR, reaction
47 (1)) to hydrogen peroxide.



49 Nonetheless, the development of metal-free electrocatalysts derived from cheap and
50 environmentally friendly biomasses for ORR promotion has become a relevant topic,⁶ especially
51 within the context of circular economy. Carbon-based materials combine low cost and
52 environmental impact with high stability, features that make them suitable candidates for H₂O₂
53 generation in actual devices.⁷⁻⁹ Likewise, porous carbonaceous materials with N-¹⁰ or S-
54 containing¹¹ surface functionalities, or both,¹²⁻¹⁴ have shown promising electroactivity and
55 selectivity for reaction (1).

56 To obtain nitrogenated carbon catalysts with high ORR performance and selectivity, three
57 aspects need to be taken into account: (a) the type of N species, which determine the selectivity
58 and the activity of the reaction sites;¹⁰ (b) the architecture, in particular the surface area and porous
59 structure; and (c) the degree of graphitization, which determines the conductivity and stability of
60 the material. Both, experimental studies and theoretical calculations, confirmed that the ORR
61 activity of nitrogen-doped carbon materials originates from the charge delocalization of the carbon
62 atoms due to nitrogen incorporation, which facilitates oxygen adsorption and reduction. The
63 activity was reported to be highly dependent on the doping level and types of nitrogen atoms.^{10,12}
64 Nitrogen functional groups can be pinned onto the carbon surface following three different
65 strategies: (i) post-functionalization of undoped carbon, by means of ion implantation,¹⁵ ball
66 milling¹⁶ or grafting;¹⁷ (ii) simultaneous carbonization and doping, feasible via activation
67 procedures like ammonia pyrolysis;¹⁸ and (iii) use of a suitable carbon-nitrogen precursor to carry
68 out the pyrolysis process.^{12,13,19-22} Such precursors are N-containing organic molecules or

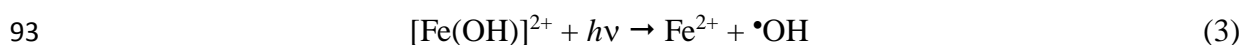
69 polymers, such as 1,10-phenanthroline and polyaniline, but a plethora of examples starting from
70 biomasses can be found in the literature.^{5,23} Examples of transformation of N-containing biomass
71 into N-doped carbons for ORR include the use of cellulose, algae, flowers, fruits or bamboo.²⁴⁻³²

72 Considering any new carbonaceous electrocatalyst of interest, a simple cathode for massive
73 H₂O₂ production could be prepared by coating a three-dimensional substrate, such as a carbon
74 felt.^{20,33} This type of substrate enhances the hydrodynamics and mass transport of oxygen,³⁴ which
75 must be continuously dissolved in the aqueous solution by conventional^{33,35,36} or advanced³⁷
76 means. Alternatively, the incorporation of an air chamber to the reactor enables a much greater
77 oxygen concentration, thereby enhancing the H₂O₂ production.⁵ In such setup, either oxygen or air
78 would be fed to a hydrophobized microporous³⁸ or macroporous³⁹ substrate coated with the
79 electrocatalyst, being the ensemble called gas-diffusion electrode (GDE).

80 Hitherto, GDEs for H₂O₂ electrogeneration have been mostly manufactured using carbon
81 black produced via hydrocarbon pyrolysis,^{2,40-44} although other carbon materials like carbon
82 nanotubes^{3,4,45} or graphene⁴⁶ have also been employed. Fewer insight has been provided on
83 mesoporous carbon,^{12,13} whose ordered structure can minimize the residence of the generated H₂O₂
84 in the reaction zone.⁴⁷ Currently, the H₂O₂ production with GDEs finds its main application in
85 water treatment, especially for the degradation of hardly biodegradable organic pollutants through
86 advanced oxidation processes like electro-Fenton (EF) ones.⁴⁸⁻⁵⁰ In EF, the very reactive •OH
87 generated from Fenton's reaction (2) allows overcoming the limitations of conventional
88 technologies.⁵¹



90 An upgraded version of EF involves the concomitant exposure of the treated solution to UVA
91 light, originating the photoelectro-Fenton (PEF) process,⁵² whose main contribution is the photo-
92 induced Fe²⁺ regeneration as follows, favoring its immediate consumption via reaction (2):^{3,4,53,54}



94 N-doping of carbons to manufacture GDEs suitable for EF or PEF has been scarcely studied.
95 The presence of N increased the electroactivity of a cathode composed of carbon nanotubes coated
96 with graphene, ending in a faster degradation of dimethyl phthalate by EF.⁵⁵ N-doped carbon
97 prepared from 1H-1,2,4-triazole-3,5-diamine showed the highest ORR rate among other materials,
98 yielding a faster EF degradation of sulfathiazole.⁵⁶ The number of studies is even lower in the case
99 of biomass-derived carbons for H₂O₂ production with GDEs. As far as we know, Liao et al.⁵⁷ have
100 reported the only article on the topic, which discusses the performance of a GDE made of N-doped
101 mesoporous carbon coming from bean dregs. They simply used the cathode to generate H₂O₂ for
102 formaldehyde degradation, which means that the investigation of biomass-derived GDEs in EF
103 and PEF is a quite unexplored field.

104 Aiming to conceive more eco-friendly EF and PEF approaches for water treatment, herein
105 chitosan has been chosen as a suitable N-containing biopolymeric gel precursor for preparing
106 mesoporous carbon material (MC-C) in virtue of its biocompatibility, biodegradability and high N
107 content (7.1%). It is worth noticing that raw chitosan has been included in some EF systems so
108 far, mainly in the form of composite with iron species to promote heterogeneous Fenton's
109 reaction,⁵⁸ although as far as we know it was never incorporated in the form of carbon powder.
110 Chitin and its derivative chitosan are among the most abundant organic compounds in nature,
111 easily obtained from crustacean shells and further purified,⁵⁹ which makes them adequate for a
112 wide range of applications.⁶⁰ The physicochemical characterization of the synthesized catalysts
113 was performed by different techniques. Their ability to electrogenerate H₂O₂ was assessed by
114 means of cyclic voltammetry (CV) and linear sweep voltammetry (LSV), as well as via bulk
115 electrolyses. In addition, the comparative degradation ability of the two catalysts was investigated
116 by treating 150 mL of acebutolol solutions at acidic pH. The selection of the contaminant is
117 justified by the growing complexity of global water pollution associated to pharmaceuticals due
118 to both, diversity of target molecules and large volume of effluents. Standard water treatments are

119 frequently unable to ensure total detoxification, which urges more advanced technologies.⁶¹ The
120 hypotensive and antiarrhythmic acebutolol is excreted mostly in its unmetabolized form and hence,
121 it is accumulated alongside its intermediates in surface water,⁶² as well as in sewage treatment
122 plant effluents, evidencing a low biodegradability.⁶³ This represents a serious threat, since
123 acebutolol is toxic for aquatic organisms⁶⁴ and causes lupus-like syndrome.⁶⁵ Herein, the PEF
124 process was assessed for the first time for acebutolol remediation.

125 MATERIALS AND METHODS

126 **Chemicals.** Chitosan (> 98%, Aldrich for Life Science) and 1,10-phenanthroline (> 99.5%,
127 TCI) were used as received. Other reagents employed for the synthesis of catalysts were: pure
128 ethanol (> 99.8%, Fluka), Nafion[®] (5 wt % in a mixture of lower aliphatic alcohols and water,
129 Sigma Aldrich), acetic acid (> 99.8%, Sigma Aldrich), acetone (> 99.5%, Sigma Aldrich), H₂SO₄
130 (95%, Fluka) and NaOH (> 99%, VWR). Each catalyst was mixed with ethanol (96%, Panreac)
131 and polytetrafluoroethylene (PTFE, 60 wt % solution, Sigma-Aldrich), following the procedure
132 described below, to manufacture a GDE. The specific reagents required for the bulk electrolytic
133 trials were: acebutolol (N-[3-acetyl-4-[2-hydroxy-3-(propan-2-
134 ylamino)propoxy]phenyl]butanamide) in the form of a hydrochloride salt (C₁₈H₂₈N₂O₄·HCl, CAS
135 number 34381-68-5, Sigma-Aldrich), Na₂SO₄ (anhydrous, Merck) and FeSO₄·7H₂O (Panreac).
136 Acetonitrile and KH₂PO₄ used to quantify acebutolol, as well as CH₂Cl₂ employed in gas
137 chromatography-mass spectrometry (GC/MS), were of high performance liquid chromatography
138 (HPLC) grade from Panreac. Other chemicals needed for the analytical procedures described
139 below included: Ti(IV) oxysulfate (technical grade, Sigma-Aldrich) to determine the H₂O₂
140 concentration; phenol (99.5%, Sigma-Aldrich), sodium nitroprusside dehydrate (Merck) and
141 ethylenediaminetetraacetic acid disodium salt dehydrate (99%, Alfa Aesar) to quantify ammonium

142 ion. Ultrapure water for synthesis, manufacture, electrolysis and analysis was obtained from a
143 Millipore Milli-Q system (Merck).

144 **Synthesis of the catalysts.** The synthesis of carbon electrocatalysts from chitosan was
145 accomplished by modifying a procedure reported in the literature.^{66,67} Chitosan was employed as
146 source of both carbon and nitrogen since it is composed of repeating units of randomly distributed
147 β -(1 \rightarrow 4)-linked D-glucosamine and N-acetyl-D-glucosamine. The synthesis consisted in the
148 formation of a chitosan hydrogel followed by the removal of the solvent by freeze drying and the
149 pyrolysis of the resulting material. The hydrogel synthesis involved the dispersion of the chitosan
150 powder (1.8 g) in an acetic acid solution (100 mL, 2 vol %), and the resulting mixture was
151 vigorously stirred to ensure the complete solubilization. In fact, at acidic pH, the amine groups in
152 the N-acetyl- β -(D)-glucosamine moiety are protonated to ammonium (pK_a (-NH₃⁺) = 6.3),⁶⁸ which
153 disrupts the hydrogen bonds between the polymer chains and leads to the collapse of the chitosan
154 crystalline structure and to the solubilization of the polymer in water, finally obtaining a
155 transparent, homogeneous and viscous solution (Figure 1a). The hydrogel formation then occurred
156 by slowly pouring a 1 M NaOH solution to the chitosan viscous solution. Since the diffusion of
157 OH⁻ ions is slow, the gelation was not instantaneous, but the gelation front advanced at the
158 hydrogel/NaOH solution interphase as the hydroxide ions diffused through the chitosan solution
159 (Figure 1b). The alkaline pH favored the deprotonation of the -NH₃⁺ groups, promoting the
160 interaction between polymeric chains by hydrogen interactions along with the incorporation of the
161 solution between the chitosan chains, yielding the hydrogel. Once the gelation was concluded, the
162 hydrogel was repetitively rinsed with water until a neutral pH was reached (Figure 1c).
163 Subsequently, the gel was cut into small pieces with a chisel and freeze-dried to remove all water.
164 The dried gel (Figure 1d) was then thermally treated in a two-step pyrolysis procedure.²⁸ The gel
165 was heated in a Carbolite tubular furnace with a 75 sccm (standard cm³/min) N₂ flux at 100 °C for
166 1 h, after which the temperature was raised up to 400 °C at a rate of 5 °C/min and kept at that

167 temperature for 2 h. The resulting powder (0.72 g) (Figure 1e), was ground by vibro-milling
168 (Retsch MM 400, four steps of 4 min/10–25 Hz) (Figure 1f) and eventually re-pyrolyzed at 900
169 °C for 2 h under nitrogen atmosphere (Figure 1g). The obtained carbon powder (MC-C, 0.52 g)
170 was washed with water and ethanol, dried at 80 °C overnight and ground by vibro-milling.

171 Alternatively, phenanthroline was used as secondary source of nitrogen and was added (0.3
172 g) to the initial chitosan/acetic acid solution, and the mixture was stirred and allowed to gel after
173 the addition of NaOH solution. This procedure allowed to finely disperse phenanthroline all over
174 the hydrogel, avoiding the preferential functionalization of the resulting carbon catalyst and
175 formation of a heterogeneous material. It is worth noting that part of the added phenanthroline was
176 washed away during the neutralization procedure with deionized water, as was confirmed by the
177 presence of the typical UV-Vis adsorption pattern of phenanthroline in the washing water (Figure
178 S1). Conversely, phenanthroline was detected only in traces in the water extracted during the
179 freeze-drying procedure (Figure S1). The resulting dried gel was subjected to a double pyrolysis
180 at 400 and 900 °C, followed by ball milling as in the previous case. The catalyst obtained starting
181 from 1,10-phenanthroline and chitosan will be denoted from now on as N-MC-C.

182 **Physicochemical characterization of catalyst powders.** Brunauer-Emmett-Teller (BET)
183 analysis, isotherm and pore distribution were performed via nitrogen adsorption-desorption at 77
184 K using the Micromeritics ASAP2020. The surface area was determined from the desorption curve
185 in a multipoint BET analysis, whereas the pore distribution was analyzed with a slit/cylindric pore
186 NLDFT equilibrium model. Elemental analysis (EA) was carried out using a Thermo Scientific
187 Flash 2000 device. Transmission electron microscopy (TEM) images were obtained by using a
188 FEI Tecnai G2 transmission electron microscope operating at 100 kV. X-ray photoemission
189 spectroscopy (XPS) measurements were performed in an UHV chamber (base pressure $< 5 \times 10^{-10}$
190 mbar), equipped with a double anode X-ray source (omicron DAR-400), a hemispherical electron
191 analyzer (omicron EIS-125) at room temperature, using non-monochromatized Mg-K α radiation

192 ($h\nu = 1253.6$ eV) and a pass energy of 50 and 20 eV for the survey and the single spectral windows,
193 respectively. To perform the XPS measurements, 2.5 mg of carbon powders were dispersed in 1
194 mL of ethanol and then sonicated for 10 min in order to obtain good dispersions; the solutions
195 were then drop-casted onto polycrystalline copper (with a diameter of 6 mm). Raman scattering
196 experiments were conducted with a DXR Raman microscope system (Thermo Fisher Scientific),
197 with a 532 nm laser as the photoexcitation source. The size of the laser spot at the sample was
198 about 25 μm and the power at the sample was 0.1, 0.5 or 1.0 mW.

199 **Electrochemical characterization of catalyst powders.** CV and LSV analysis at a rotating
200 ring disk electrode (RRDE, Metrohm; 5 mm diameter glassy carbon (GC) disk + Pt ring, with a
201 collection efficiency of 25% were performed in both Ar-purged and O₂-saturated 0.0005 M H₂SO₄
202 + 0.050 M Na₂SO₄ solutions by using an Autolab Model 101 N potentiostat/galvanostat. A three-
203 electrode configuration was used, consisting in a GC disk (geometric area of 0.196 cm²) as working
204 electrode, a graphite rod as counter electrode, and a reversible hydrogen electrode (RHE) as
205 reference electrode. The latter was freshly prepared before each experiment and consisted in a Pt
206 wire mesh sealed to the closed end of a capillary glass tube and refilled with the electrolyte solution
207 from the other open end. H₂ was directly electrogenerated at the Pt wire mesh so that half of the
208 Pt mesh was exposed to the H₂ bubble confined between the electrolyte solution and the closed
209 end of the capillary.⁶⁹

210 The MC-C and N-MC-C catalysts were characterized as thin films prepared by drop-casting
211 an ink (20 μL) of the corresponding carbon powder on a GC disk. Before the drop casting, the GC
212 was polished to a mirror finish with Struers silicon carbide papers of decreasing grain size (grit:
213 500, 1000, 2400, 4000) followed by diamond paste (3-, 1-, 0.25-mm particle size) and repeatedly
214 washed and sonicated (10 min each time) in ethanol for removing all contaminants.

215 All the electrochemical assays were carried out with an optimized loading of 0.6 mg/cm². The
216 electrolyte was purged with Ar before each measurement, whereas for the ORR test, high-purity

217 O₂ gas was bubbled through the electrolyte for at least 1 h to ensure O₂ saturation. The number of
218 electrons transferred during ORR was determined by the RRDE technique. Before data acquisition,
219 the electrocatalysts were first activated by cycling the electrode in the solvent potential window at
220 200 mV/s until obtaining a stable cyclic voltammogram.

221 **Fabrication of the gas-diffusion cathodes.** A GDE was prepared from each type of
222 synthesized powder using the spraying method.^{3,4} An appropriate amount (0.1 g) of MC-C or N-
223 MC-C powder was ultrasonically dispersed with PTFE and ethanol for 45 min to obtain an ink (~
224 20 mL). Carbon cloth (~8 cm² geometric area, BASF B1ASWP), degreased in ethanol and then
225 dried at 60 °C for 3 min, was used as substrate to spray the ink. An air-brush gun fed with N₂ gas
226 was employed to apply several layers, until the ink was finished. After each layer, the material was
227 dried at 60 °C for 3 min and then weighted. The sample was pressed at 2 ton for 45 s, annealed at
228 400 °C for 60 min under nitrogen atmosphere and finally cooled down at ambient conditions. The
229 overall weight increment in each GDE was ~80 mg (i.e., MC-PTFE loading of 9.7 mg/cm²). In
230 order to use any new GDE, it was first activated conducting a galvanostatic polarization in 0.050 M
231 Na₂SO₄ at pH 3.0 for 60 min.

232 The morphological characteristics of the GDEs before and after use were assessed by
233 scanning electron microscopy (SEM) using a JEOL JSM-7100F field-emission microscope
234 equipped with an energy-dispersive X-ray spectroscopy analyzer. All images were obtained at a
235 voltage of 20.0 kV.

236 **Bulk electrolyses and analyses.** Electrolytic trials were conducted for several hours in the
237 absence and presence of pollutant to evaluate the H₂O₂ electrogeneration ability of GDEs and the
238 degradation performance of different processes, respectively. In all these trials, a given GDE (3
239 cm²) prepared as described above was placed in a tubular polypropylene housing that received
240 compressed air pumped at 36 L/h, and such cathode was connected to the anode (3 cm²). The latter
241 consisted in a RuO₂-based plate (i.e., dimensionally stable electrode, DSA) from NMT Electrodes

242 in most of the trials, although boron-doped diamond (BDD) from NeoCoat was also employed in
243 some cases. The distance between the anode and cathode was 1.0 cm. The assays were performed
244 in an undivided glass cell, which was placed onto a magnetic stirrer operated at 700 rpm and
245 contained 150 mL of solution at pH 3.0 and thermostated at 25 °C. All treatments were made in
246 the presence of 0.050 M Na₂SO₄ as background electrolyte and, when required, acebutolol was
247 added at a concentration of 0.046 mM, i.e., 10 mg/L total organic carbon (TOC). In EF and PEF,
248 Fe²⁺ catalyst was added before starting the electrolysis at a concentration of 0.50 mM. In the photo-
249 assisted treatment, the solution was illuminated with UVA photons from a 6 W Philips tubular
250 lamp ($\lambda_{\text{max}} = 360$ nm, irradiance of 5 W/m²) placed above the reactor near the liquid surface.
251 Constant current was always supplied by an Amel 2051 potentiostat-galvanostat, whereas the cell
252 voltage (E_{cell}) was continuously displayed by a Demestres 605 multimeter.

253 When required, the pH of solutions was measured with a Crison GLP 22 pH-meter.
254 Colorimetric analysis of the complex formed between Ti(IV) and H₂O₂ was made using an Unicam
255 UV/Vis spectrophotometer set at $\lambda_{\text{max}} = 408$ nm, with solutions thermostated at 25 °C. The content
256 of ammonium ion was measured with the same equipment, according to the indophenol blue
257 method, at $\lambda_{\text{max}} = 630$ nm.³ TOC was determined using the non-purgeable organic carbon (NPOC)
258 mode of a Shimadzu TOC-VCNS analyzer. Acebutolol concentration was determined by reversed-
259 phase HPLC, with a Waters 600 apparatus equipped with a C18 column at 35 °C connected to a
260 photodiode array detector. A well-resolved peak appeared at retention time of 4.6 min ($\lambda = 235$
261 nm). The mobile phase was a mixture of acetonitrile (40 vol %) and water (10 mM KH₂PO₄ at pH
262 3.0, 60 vol %), circulating at 1.0 mL/min. Trials were made in duplicate and injections in triplicate
263 to correctly assess the drug disappearance. Hence, average values are shown in the figures
264 alongside the error bars (95% confidence level). The final carboxylic acids were quantified as
265 previously reported, by injecting the samples in the same chromatograph but equipped with an

266 Aminex column.³⁹ Before each analysis, the samples were conditioned by filtration with PTFE
267 filters (13 mm × 0.45 μm, Whatman).

268 Current efficiency (CE) values during the H₂O₂ electrogeneration were calculated from the
269 applied charge, according to the Eq. (4):⁵

$$270 \quad \text{CE (in \%)} = \frac{2F[\text{H}_2\text{O}_2]V}{1000M(\text{H}_2\text{O}_2)Q} 100 \quad (4)$$

271 where 2 is the stoichiometric number of electrons transferred for ORR to form H₂O₂, *F* is the
272 Faraday's constant (96,487 C/mol), [H₂O₂] is the concentration of accumulated H₂O₂ (mg/L), *V* is
273 the volume of the treated solution (L), 1000 is a conversion factor, *M*(H₂O₂) is the molecular
274 weight of H₂O₂ (34 g/mol), and *Q* is the charge consumed during the electrolysis.

275 The mineralization current efficiency (MCE) for each trial at a given electrolysis time *t* (in h)
276 and applied current *I* (in A) was calculated as follows:⁷⁰

$$277 \quad \text{MCE (in \%)} = \frac{nFV(\Delta\text{TOC})_{\text{exp}}}{4.32 \times 10^7 mIt} 100 \quad (5)$$

278 where *n* accounts for the number of electrons consumed for the overall mineralization of the drug,
279 Δ(TOC)_{exp} is the observed TOC decay (in mg/L), 4.32×10⁷ is a conversion factor (= 3600 s/h ×
280 12000 mg C / mol) and *m* is the number of carbon atoms of acebutolol (18 atoms).

281 The specific energy consumption per unit TOC mass (EC_{TOC}) was obtained as follows:⁷⁰

$$282 \quad \text{EC}_{\text{TOC}} \text{ (in kWh / g TOC)} = \frac{E_{\text{cell}}It}{V(\Delta\text{TOC})_{\text{exp}}} \quad (6)$$

283 where *E*_{cell} is the average cell voltage (in V).

284 The primary reaction products were identified by GC/MS. For this, samples collected after
285 selected electrolyses were prepared by liquid-liquid extraction with CH₂Cl₂ as organic solvent.³⁶

286 The GC/MS analysis was made in electron impact mode at 70 eV with an Agilent Technologies
287 system: a 6890N gas chromatograph with a 7683B series injector was connected to a 5975 mass
288 spectrometer. Non-polar Teknokroma Sapiens-X5.ms and polar HP-INNOWax columns, both
289 with dimensions of 0.25 μm, 30 m x 0.25 mm (i.d.), were used. When using the former, the

290 conditions were: temperature ramp starting at 36 °C for 1 min and increasing up to 320 °C at 5
291 °C/min (hold time 10 min); the temperature of the inlet, source and transfer line was 250, 230 and
292 300 °C, respectively, and the analyses were made by splitless (0.7 min) injection, with run time of
293 67.80 min. When the latter was employed, the conditions were analogous but the final temperature
294 was 250 °C, with 250 °C as the temperature of the transfer line and 93.80 min as run time.

295 **RESULTS AND DISCUSSION**

296 **Characterization of synthesized catalysts.** Nitrogen-doped carbon is a term commonly
297 accepted by the scientific community to classify carbon powder containing nitrogen functional
298 groups. In the present paper, the pyrolysis of the pure dried chitosan hydrogel as well as of the
299 dried hydrogel embedding 1,10-phenanthroline yielded two samples, denoted as MC-C and N-
300 MC-C, respectively. The three-dimensionality of chitosan hydrogels allows the fine dispersion of
301 additives inside the gel structure, which can improve the functionalization of the carbon structure
302 and enhance the textural properties resulting from thermal treatment. A double step pyrolysis, at
303 400 and 900 °C, was employed for decreasing the burn off of the material, and the resulting black
304 powder was characterized after ball milling.

305 Figure 2 highlights the TEM images of the two types of carbon powder at different
306 magnifications. The structure of the MC-C sample is characterized by the overlay of compact
307 carbon lamellae (Figure 2a-c), in which any pore structure is clearly visible. The morphological
308 features of MC-C are in part superimposable with those of N-MC-C (Figure 2d-f). However, the
309 lamellar structures are in this latter case characterized by a less packed and compact structure,
310 presenting more jagged edges.

311 Table 1 reports the elemental analyses of the synthesized carbons. MC-C, prepared by
312 employing the sole chitosan as both carbon and nitrogen precursor, shows a nitrogen fixation of
313 almost 7%, as evidenced by the CHN analysis. Then, it must be clear that the notation MC-C

314 simply means that an additional N-rich precursor was not employed during the synthesis. Also the
315 oxygen content, which in first approximation can be calculated as the residual mass percentage, is
316 sensitively high (~16%). Therefore, it is reasonable to infer that the MC-C carbon was rich in
317 nitrogen and oxygen functional groups, which are extremely important as active sites to
318 electrocatalyze some reactions as well as for increasing the wettability of the electrode material
319 during the electrocatalytic tests. A close comparison with the values obtained for the N-MC-C
320 sample (Table 1) allows noticing that, when 1,10-phenanthroline was used as a secondary nitrogen
321 source during the chitosan hydrogel preparation, the resulting carbon had a higher carbon and
322 hydrogen content but a lower nitrogen (< 5%) and oxygen (< 15%) percentage as compared to
323 MC-C. We can presume that the pyrolysis of 1,10-phenanthroline generates small gaseous
324 molecules such as NO, NO₂ and CO, which may act as oxidizing agents able to react with the
325 amorphous part of the carbon structure, i.e., the sp³ carbon atoms bonded to nitrogen and oxygen
326 functional groups, while the more graphitic sp² are preserved, being less prone to react with in situ
327 generated reactive species.

328 The nitrogen adsorption/desorption isotherms at 77 K for MC-C belong to the type II, which
329 is characteristic for non-porous or macroporous materials having a relatively small external surface
330 (Figure 3a). However, the TEM pictures do not show the presence of macropores (Figure 2a-c),
331 which would correspond to pores with diameters wider than 50 nm, being reasonable to conclude
332 that MC-C can be classified as a non-porous carbon material. The sample showed a non-reversible
333 desorption behavior with an open hysteresis loop classified as type-H4, which is often attributed
334 to narrow slit-like pores.⁷¹ The very low value of mesopore volume (Table 1) can be associated to
335 the collapse of the mesopore structure during the gas desorption or to the entrapment of N₂ gas in
336 narrow pores formed between two carbon foil, which could also explain the open hysteresis.⁷²
337 However, the experiment was repeated a second time, showing the very same behaviour and hence,
338 the first hypothesis of a carbon structure collapse can be disregarded, being the irreversible

339 confinement of N₂ molecules between carbon foils the most probable explanation. The
340 extrapolated BET surface area was as small as $A_{\text{BET}} = 6 \text{ m}^2/\text{g}$. Figure 3a also reports the type II
341 nitrogen adsorption/desorption isotherm for N-MC-C. Similarly, in this case the open hysteresis
342 loop can be classified as type-H4, and an analogous conclusion can be drawn regarding the open
343 hysteresis. It is worth noting that the presence of 1,10-phenanthroline in the dried hydrogel affected
344 very positively the BET surface area, which was enhanced up to $63 \text{ m}^2/\text{g}$ in N-MC-C. In this case,
345 the more pronounced uptake at low p/p^0 can be associated with an enhanced adsorbent-adsorbate
346 interaction in narrow micropores.⁷² The large area was directly related to the increased porosity
347 (Table 1), which was mainly due to micropores ($d < 2 \text{ nm}$, $V_{\mu} = 0.021 \text{ cm}^3/\text{g}$), along with a
348 mesopore fraction ($2 \text{ nm} < d < 50 \text{ nm}$, $V_{\text{meso}} = 0.017 \text{ cm}^3/\text{g}$). The formation of micropores and
349 mesopores or the simple opening of clogged pores can be associated to the reaction of NO, NO₂
350 and CO gaseous species, generated during the pyrolysis of 1,10-phenanthroline with the carbon
351 structure while CO₂ is released.^{73,74} The pronounced uptake at high p/p^0 is indicative of the
352 presence of macropores ($d > 50 \text{ nm}$), which cannot be in any case evaluated in terms of pore
353 volume and area by the adopted model. However, the presence of a more open and accessible
354 structure finds a confirmation in the TEM images described above. Note that the higher porosity
355 of N-MC-C as compared to MC-C is also evident from the greater total volume (V_{TOT}) summarized
356 in Table 1 ($0.065 \text{ vs } 0.013 \text{ cm}^3/\text{g}$, respectively). Therefore, while MC-C is a non-porous carbon
357 material, N-MC-C can be classified as a carbon material with low porosity and textural properties
358 similar to those of commercial carbon black ($A_{\text{BET}} = 67 \text{ m}^2/\text{g}$, $V_{\mu} = 0.015 \text{ cm}^3/\text{g}$, $V_{\text{meso}} = 0.137$
359 cm^3/g).

360 The MC-C and N-MC-C samples were characterized by Raman spectroscopy, which allows
361 elucidating the possible differences in the order and aggregation degree of sites with sp²
362 hybridization and, more in general, informs about the amorphous and graphitized degree of a
363 carbon material (Figure 3b).⁷⁵ Soft and hard carbon materials typically show two main bands: the

364 D1 (disorder) at $\sim 1350\text{ cm}^{-1}$ and the G (graphitic) at $\sim 1595\text{ cm}^{-1}$, as corroborated also for the MC-
365 C and N-MC-C samples. The 2D peak at $\sim 2680\text{ cm}^{-1}$, ascribed to an out-of-plane vibration mode
366 and generally present in graphite, single layer graphene or highly graphitized carbon is missing,
367 but a broad band between 2400 and 3000 cm^{-1} is present in both samples. The spectra
368 deconvolution revealed the existence of four bands after baseline subtraction and normalization to
369 the maximum of the G band. The two further bands D4 and D3 account for polyenes or ionic
370 impurities and amorphous carbon, respectively. The resulting I_{D1}/I_G band intensity ratio was close
371 to one for both samples (Figure S2). According to the Tuinstra-Koenig model, the following
372 relationship is satisfied:⁷⁶

$$373 \quad \frac{I(D1)}{I(G)} = \frac{C(\lambda)}{L_a} \quad (7)$$

374 where $C(\lambda) = -12.6 + (0.033\lambda) \approx 50\text{ \AA}$, ($\lambda = 532\text{ nm}$) and thus, the band intensity ratio can be used
375 to determine the size of the graphite nanocrystallites, which was 4.7 and 5.0 nm for MC-C and N-
376 MC-C, respectively. A close comparison of the two spectra allows observing two main differences:
377 (i) the D1 intensity was higher in the MC-C spectrum than in the N-MC-C one; and (ii) the valley
378 (minimum intensity point) between the D1 and G band was higher in MC-C than in N-MC-C. Both
379 characteristics lead to conclude that N-MC-C was more graphitized than MC-C. This can be further
380 confirmed by considering the D3 band area, responsible for the amorphous carbon, which resulted
381 lower in N-MC-C (37.6 normalized counts) with respect to MC-C (41.6 normalized counts).

382 The XPS analyses confirmed the presence of nitrogen, oxygen and carbon in both samples
383 (Table 1 and Figure S3). The higher percentage of oxygen in MC-C with respect to N-MN-C
384 confirmed that the introduction of 1,10-phenanthroline contributed to a greater graphitization.
385 Figures 3c and 3d and Table S1 show the C $1s$ XPS data of the analyzed samples. The main carbon
386 component at binding energy (B.E.) = 284.4 eV was attributed to C- sp^2 .^{12,15,77} At higher B.E.
387 (285.4 eV) the peak corresponded to the sp^3 C-C bond. The higher sp^2/sp^3 ratio in N-MC-C as
388 compared to MC-C sample is in agreement with the observations from Raman analysis (Table S1).

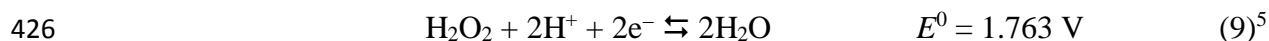
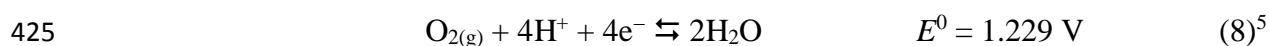
389 There was also a substantial amount of carbon bound to nitrogen or/and to oxygen (peak at 286.4
390 eV). Peaks at B.E. higher than 286.4 eV were due to carbon species bound to oxygen (C=O, O-
391 C=O, O-C(O)-O) or to both oxygen and nitrogen (C-N=O, N-C=O). Peak at 291.1 eV was due to
392 shakeup satellite ($\pi\text{-}\pi^*$).

393 The N 1s spectra and their deconvolution in single peaks are reported in Figure 3e and 3f. The
394 elemental analysis determined from XPS data indicates a weight percentage of surface nitrogen of
395 2.6% and 3.4% for MC-C and N-MC-C, respectively, in contrast to the above mentioned CHN
396 analysis, which is a bulk sensitive technique that detected a greater nitrogen content in the MC-C
397 sample (Table 1). Therefore, it is reasonable to assert that in the MC-C catalyst the nitrogenated
398 functional groups are mostly confined in the bulk of the carbon material, whereas the pyrolysis of
399 chitosan hydrogel carried out in the presence of 1,10-phenanthroline leads to a higher surface
400 nitrogen content, thereby providing a higher number of available active sites. This is a key finding
401 that justifies the presence of the additional N-containing precursor during the synthesis, since the
402 resulting surface N-functionalities are expected to induce a higher H₂O₂ electrogeneration. The
403 speciation of the singular N 1s components evidences the presence of pyridinic, imine, amine and
404 pyrrolic nitrogen (Table 1). Graphitic and nitrogen oxide components were also present in lower
405 percentage.^{12,28} It is interesting to observe that the pyrrolic component significantly increases in
406 N-MC-C. Both, pyridinic and pyrrolic nitrogen are beneficial groups since they increase the carbon
407 activity versus ORR and, more in particular, it is reported that pyrrolic nitrogen favors the bi-
408 electronic reduction of oxygen to hydrogen peroxide.^{17,27,30}

409 **Electroactivity of chitosan-derived catalysts.** MC-C and N-MC-C were both
410 electrochemically characterized in 0.0005 M H₂SO₄ + 0.050 M Na₂SO₄ as background electrolyte,
411 aiming to determine their activity regarding the oxygen reduction and their corresponding
412 selectivity towards the H₂O₂ generation. Figure 4a and 4b compare the electrochemical behaviour
413 in Ar-purged and O₂-saturated electrolyte using MC-C and N-MC-C electrocatalyst, respectively.

414 No redox signal appeared in the purged medium, whereas both materials showed electroactivity
 415 versus ORR, as deduced from the well-defined reduction peak that was visible in the presence of
 416 dissolved O₂. The O₂ reduction peak potential using the N-MC-C (0.525 V vs RHE) was 190 mV
 417 more positive than that obtained with the MC-C (0.335 V vs RHE). The smaller overpotential
 418 required using the N-MC-C informs about its higher activity with respect to MC-C, which can be
 419 attributed to the aforementioned greater content of nitrogen groups on the N-MC-C surface (Table
 420 1).

421 The ORR may proceed through two pathways: one is a direct four-electron route, which
 422 consists in the direct O₂ reduction to water without involvement of H₂O₂ (reaction (8)); the other
 423 is a two-step two-electron route in which H₂O₂ is first formed as intermediate via the 2-electron
 424 reaction (1), and then it can potentially undergo further reduction via reaction (9).



427 The selectivity towards the formation of H₂O₂ was evaluated by LSV using an RRDE in O₂-
 428 saturated electrolyte at a scan rate of 10 mV/s and rotation speed (ω) of 1600 rpm. During RRDE
 429 experiments, the potential of the disk electrode was swept to cathodic values, so that the
 430 appearance of a current (I_D) in the linear voltammogram using MC-C and N-MC-C confirmed the
 431 reduction of O₂, since this was the sole reducible species present in solution (Figure 4c and 4d,
 432 respectively). Due to the electrode rotation, the current increased until a limiting value (I_{lim}) was
 433 attained, which depended primarily on the electrode ω , the dissolved O₂ concentration (C_{O_2}), the
 434 O₂ diffusion coefficient (D_{O_2}), the solvent kinematic viscosity (ν) and the number of transferred
 435 electrons (n), according to the Koutecky–Levich equation:¹²

$$436 \quad \frac{1}{I} = \frac{1}{I_k} + \frac{1}{I_{lim}} = \frac{1}{nFAkC_{O_2}} + \frac{1}{0.62nFA(D_{O_2})^{2/3}\nu^{-1/6}C_{O_2}\omega^{1/2}} \quad (10)$$

437 where I_k is the kinetic current and I_{lim} is the diffusion limiting current. In such setup, the electrode
 438 rotation pushes the electrogenerated species toward the ring electrode, which is polarized at a fixed

439 potential ($E_{\text{ring}} = 1.50$ V vs. RHE), at which H_2O_2 can be easily reoxidized to H_2O . If H_2O_2 is
 440 generated at the disk electrode, it can then be detected at the ring electrode and a ring current (I_R)
 441 can be recorded in the case of MC-C and N-MC-C (Figure 4c' and 4d', respectively). The number
 442 of exchanged electrons can be in principle be determined by plotting the reciprocal of the total
 443 instant current (I^{-1}) versus $\omega^{-1/2}$, whenever all the other specific parameters are known. However,
 444 this procedure can bring to misleading results when, for example, the catalyst layer is not
 445 homogeneous and sufficiently thin. Alternatively, the number of electrons involved in the ORR
 446 and the yield of hydrogen peroxide ($y_{\text{H}_2\text{O}_2}$) can be obtained from the I_D (in A) and I_R (in A) values
 447 measured at the RRDE, according to Eqs. (11) and (12):⁷²

$$448 \quad n = \frac{4I_D}{I_D + I_R/N} \quad (11)$$

$$449 \quad y_{\text{H}_2\text{O}_2} = 100 \frac{2I_R}{NI_D + I_R} \quad (12)$$

450 where N is the collection efficiency ($N = 0.25$). The half-wave potential, which can be determined
 451 from the RRDE measurements, confirmed the higher catalytic activity of N-MC-C ($E_{1/2} = 0.437$
 452 V) as compared to MC-C ($E_{1/2} = 0.288$ V). Furthermore, the n values were 3.55 and 3.76, using
 453 N-MC-C and MC-C, respectively, which informs about the superior selectivity versus H_2O_2
 454 generation using the former electrode. In fact, the hydrogen peroxide yields ($y_{\text{H}_2\text{O}_2}$, Figure 4c''
 455 and 4d'') were 25% (N-MC-C) and 11% (MC-C). At first sight, these values could seem to far
 456 from $n = 2$ and $y_{\text{H}_2\text{O}_2} = 100\%$ ideally expected from an electrocatalyst that fully promotes the two-
 457 electron ORR, without further H_2O_2 reduction or simultaneous reaction (5). Nonetheless, the main
 458 goal within the context of this work was not the development of an optimum electrocatalyst for
 459 H_2O_2 electrosynthesis, but the synthesis of a potentially cheaper and more eco-friendly material to
 460 electrogenerate sufficient amounts of H_2O_2 for its direct application in Fenton-based
 461 electrochemical water treatments. Therefore, bulk electrolyses were conducted in order to assess
 462 the ability of GDEs, prepared with the synthesized MC-C and N-MC-C, to accumulate H_2O_2 .

463 A series of long galvanostatic electrolyses was carried out, at different current densities from
464 10 to 60 mA/cm², to monitor the concentration of H₂O₂ accumulated in a 0.050 M Na₂SO₄
465 solutions at pH 3.0 for 360 min. In these trials, either an MC-C- or N-MC-C-derived GDE was
466 connected to a DSA plate. As can be seen in Figure 5a, with both types of GDE, a higher
467 concentration was achieved as the electrolysis progressed, as a result of the effective ORR via
468 reaction (1). However, the initially linear trends became curves after 120-180 min, which means
469 that the processes that accounted for the H₂O₂ destruction gradually became more significant. The
470 main ones among them were: (i) the cathodic reduction via reaction (6), (ii) its oxidation at the
471 anode surface, since the cell had one single compartment, and (iii) the chemical decomposition
472 and natural disproportionation in the bulk. In fact, a final plateau could be observed in some cases,
473 especially at the highest current density of 60 mA/cm², which was reached once the H₂O₂
474 formation and destruction rates became equal. At 360 min, the concentrations attained at 10, 30
475 and 60 mA/cm² were 3.68, 9.35 and 9.75 mM using the MC-C, and 5.20 13.2 and 15.3 mM with
476 the N-MC-C. Therefore, the greatest values appeared in the cells with the N-MC-C-derived GDE,
477 regardless of the applied current density, which confirms the importance of the superficial N
478 content and the corresponding N-functionalities described above. Furthermore, the difference was
479 higher as the current density was increased (i.e., the curves with both GDEs were comparatively
480 more separated), suggesting a relatively greater stability of the N-MC-C under more aggressive
481 conditions. In fact, using the MC-C-derived GDE, the H₂O₂ contents at 30 and 60 mA/cm² were
482 very close, meaning that the stability range for that cathode was more limited. Such lower stability
483 can be related to its structural characteristics, discussed above, since a lower porosity and greater
484 compactness constitute a barrier that oxygen gas tends to break in a rough manner.

485 A comparison with reported values for H₂O₂ accumulation in similar setups that employed
486 commercial C-PTFE GDEs allows concluding that the electrogeneration performance observed in
487 this work is in the same range, in particular in the case of the GDE prepared with the N-MC-C.

488 For example, 0.59 mM H₂O₂ (vs 2.00 mM in this work) were measured at 30 min in a 0.050 M
489 Na₂SO₄ solution at pH 3.0 recirculated at a liquid flow rate of 4.4 L/min and electrolyzed at 30
490 mA/cm² in a pilot plant, whereas a content of 17 mM H₂O₂ was achieved after 360 min at 50
491 mA/cm² in a 2.5 L plant.⁷⁸ Moreover, the values represented in Figure 5a are much better than
492 those typically attained with raw carbon-felt cathodes, which have also been employed in EF and
493 PEF treatments.⁵ These results corroborate that the electroactivity of the synthesized catalysts was
494 high enough to accumulate H₂O₂, despite the partial selectivity discussed from Figure 4. For a
495 more accurate comparison with the literature, Table S2 summarizes the current efficiency, H₂O₂
496 mass production rate and H₂O₂ yield calculated for our best catalyst as well as for carbon
497 electrocatalysts employed in GDEs for water treatment. Note that, despite the moderate efficiency
498 of the new material to accumulate H₂O₂, the crucial finding of this work is that the electrogenerated
499 amount is sufficient to promote the degradation of water pollutants (see subsections below).

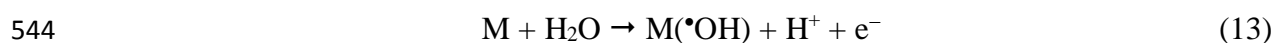
500 The presence of Fe²⁺ in the solution (i.e., EF conditions) should cause a drastic reduction in
501 the quantity of H₂O₂ accumulated due to its quick decomposition via Fenton's reaction (2). Indeed,
502 this is verified in Figure 5b, which shows similar profiles regardless of the type of cathode when
503 Fe²⁺ at a concentration of 0.50 mM was employed as catalyst. The other conditions were analogous
504 to those described in Figure 5a, at 30 mA/cm². Based on the almost identical trends to finally attain
505 a value of ~5.1 mM H₂O₂, and considering that the N-MC-C-derived GDE was proven to generate
506 more H₂O₂ than the MC-C GDE in the absence of Fe²⁺ (Figure 5a), a greater •OH production can
507 be inferred. This should entail a faster water decontamination, as will be discussed later. The result
508 obtained in these EF systems is in good agreement with that previously reported with a commercial
509 GDE in an analogous solution (< 10 mM H₂O₂ at 30 mA/cm²).⁷⁹ Similar experiments were made
510 exposing solutions with the same composition to UVA photons (PEF conditions), which caused
511 an additional reduction in the content of the oxidant accumulated at each electrolysis time until
512 reaching ~4.6 mM H₂O₂ at 360 min (Figure 5b). This is explained by the occurrence of photo-

513 Fenton reaction (3), with optimum $\lambda = 360$ nm. The continuous Fe^{2+} photo-induced regeneration
514 sustained the Fenton's reaction (2), which otherwise would be mitigated due to the gradual
515 conversion of Fe^{2+} into the much less active species $[\text{Fe}(\text{OH})]^{2+}$ (i.e., the main Fe(III) species at
516 pH 3.0).⁵

517 The current efficiencies for all the trials depicted in Figure 5a and 5b, determined from Eq.
518 (4), are illustrated in Figure S4a and S4b, respectively. In all cases, a loss of efficiency over time
519 can be observed, in agreement with the progressively lower H_2O_2 accumulation rate (i.e., the
520 curvature appearance after some minutes in Figure 5 owing to parasitic reactions). In the absence
521 of Fe^{2+} (Figure S4a), the maximum efficiency of $\sim 40\%$ (reasonably good taking into account that
522 the cell was undivided) corresponded to the treatment with N-MC-C at 10 mA/cm^2 . Worth
523 noticing, upon increase to 30 mA/cm^2 the efficiency profile was very similar, which is interesting
524 in practice to promote a faster degradation. Therefore, this current density was selected as optimum
525 for subsequent trials. When the current density was then doubled, the efficiency was almost halved,
526 confirming the detrimental enhancement of parasitic destruction reactions (i.e., cathodic and
527 anodic phenomena mentioned before). All the treatments with the MC-C-derived GDE were much
528 less efficient ($< 25\%$ during all the electrolysis). As expected from the practically overlapping
529 H_2O_2 profiles found in EF and PEF (Figure 5a), Figure S4b evidences similar trends for the
530 corresponding current efficiencies (always lower than 20%), with slightly higher values in EF
531 assays.

532 **Degradation of acebutolol using GDEs prepared with the synthesized catalysts.** For the
533 dual purpose of assessing the viability of the new GDEs to foster the degradation of organic
534 pollutants and confirming the superiority of N-MC-C electrocatalyst, solutions containing 0.046
535 mM of the β -blocker acebutolol and $0.050 \text{ M Na}_2\text{SO}_4$ at pH 3.0 were electrolyzed at the optimized
536 current density (30 mA/cm^2). In Figure 6a, the drug concentration decays under electro-oxidation
537 (EO- H_2O_2) conditions with each GDE. The degradation was quite slow in both cases, ending in

538 modest removals of 35% and 51% at 180 min using the MC-C- and N-MC-C-derived GDE,
539 respectively. Since H₂O₂ behaves as a rather mild oxidant in front of aromatic structures like that
540 exhibited by acebutolol, within the concentration ranges as that discussed from Figure 5a,⁶ the
541 partial drug disappearance can be attributed to the action of hydroxyl radicals M([•]OH) formed on
542 the anode (M) surface from reaction (13).^{40,48} The data could not be fitted well considering a simple
543 kinetic model, thus suggesting a larger complexity due to the influence of reaction products.

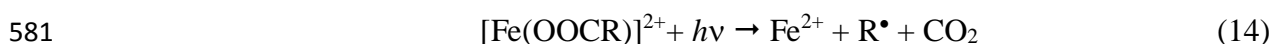


545 The degradation was drastically upgraded in the presence of 0.50 mM Fe²⁺ and UVA photons
546 under PEF conditions, as evidenced in Figure 6b. The use of the two DSA/GDE cells led to 97%
547 and 100% acebutolol removal in only 20 min using the MC-C and N-MC-C electrocatalyst,
548 respectively. Such significant acceleration of concentration decays was pre-eminently favored by
549 the generation of [•]OH upon fast decomposition of H₂O₂ via Fenton's reaction (2), which confirms
550 the explanation given on the time course of H₂O₂ (Figure 5b). The effective [•]OH production was
551 ensured by the continuous Fe(III) photoreduction from reaction (3), which yielded additional
552 amounts of [•]OH and regenerated the Fe²⁺. The replacement of the RuO₂-based anode by BDD had
553 a positive influence on the destruction of the pollutant, being more evident in the system with N-
554 MC-C, since the complete disappearance was observed at 16 min. Using MC-C, almost no change
555 in degradation percentage (98%) was reached, but the initial concentration decay became quicker.
556 This is clear in the inset of Figure 6b, which depicts the excellent linear fitting ($R^2 > 0.990$)
557 resulting from a pseudo-first-order kinetic analysis of all decays of the main figure. The pseudo-
558 first-order rate constant (k_1) increased from 0.1748 to 0.2296 min⁻¹ when the anode was changed
559 to BDD in the system with MC-C. Accordingly, higher k_1 -values were determined using N-MC-
560 C, namely 0.2583 and 0.3258 min⁻¹ with DSA and BDD, respectively. The enhancement achieved
561 with BDD anode is associated to the physisorbed nature of M([•]OH) produced from reaction (13),
562 which makes the radicals more reactive with organic molecules approaching its surface.⁴⁸

563 However, it is clear that the main factors affecting the drug degradation were those related with
564 the Fenton's reactivity, as can be deduced when comparing Figure 6a and 6b. Among such factors,
565 the benefit of the higher H₂O₂ electrogeneration with N-MC-C stands out.

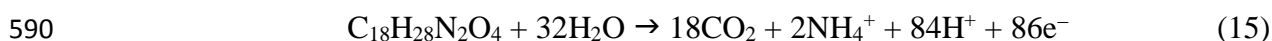
566 The performance of our biomass-derived GDEs is comparable to that reported for GDEs
567 prepared with carbon cloth coated with commercial carbon black. For example, the treatment of
568 β -blocker atenolol in 0.050 M Na₂SO₄ at pH 3.0 by PEF at 16.7 mA/cm² allowed the total drug
569 removal in 30-35 min, with $k_1 = 0.05 \text{ min}^{-1}$, thus being in the same order of magnitude.⁸⁰

570 Figure 7 highlights the time course of normalized TOC content with electrolysis time during
571 the PEF assays of Figure 6b. Unlike that observed in the previous HPLC analysis, the effect of the
572 anode nature was crucial to upgrade the mineralization of the solutions, which initially contained
573 10 mg/L TOC. Partial TOC abatements were obtained with DSA, attaining 71% and 76% at 360
574 min using MC-C and N-MC-C, respectively. A much greater decontamination was feasible upon
575 use of BDD anode, with final values of 91% and 97%. In all these systems, the effective generation
576 of •OH in the bulk sustained by reaction (3) undoubtedly contributed to the gradual degradation of
577 reaction products. However, this was insufficient because it is known that the aromatic molecules
578 are transformed into very refractory aliphatic products and their complexes with Fe(III) that resist
579 quite well the attack of •OH.^{3,4} Therefore, UVA photons played an additional role, allowing the
580 photodecarboxylation of such stable complexes:



582 As inferred from both PEF trials with DSA, the degradation power of •OH and UVA light was
583 still limited and hence, M(•OH) formed at BDD favored the electrocatalytic destruction of the
584 refractory molecules. As a result, the combination of the most effective anode and cathode yielded
585 the greater mineralization. Note that TOC decays reported for acebutolol solutions through other
586 advanced oxidation processes were lower, as in the case of photocatalysis with TiO₂ with 75%
587 mineralization at 240 min.⁸¹

588 The following overall mineralization reaction can be proposed, based on the determination of
589 inorganic ions discussed below:⁵



591 From Eq. (5), in which the total number of electrons involved was $n = 86$, the MCE values
592 calculated at a given electrolysis time for the assays of Figure 7 are represented in Figure S5a. The
593 most efficient process was PEF with the BDD/GDE cell using N-MC-C, in agreement with the
594 effective TOC removal in that system. Nonetheless, the fact that the maximum MCE was $< 15\%$
595 reflects the high refractoriness of the pollutant and its derivatives. As the treatments progressed, a
596 loss of efficiency was observed, meaning that the organic structures and complexes formed were
597 more resistant to $\bullet\text{OH}$, $\text{M}(\bullet\text{OH})$ and UVA photons than the target pollutant and its primary
598 intermediates. In addition, the side reactions that wasted the radicals and photons occurred to a
599 greater extent as the number of available organic molecules decreased. These two phenomena with
600 negative impact on the process efficiency were also detrimental in terms of energy consumption.
601 The EC_{TOC} values calculated from Eq. (6) and represented in Figure 6b were higher in trials with
602 BDD, which is mainly explained by the much greater E_{cell} as compared to those with the RuO_2 -
603 based anode (15.5 vs 8.0 V, regardless of the GDE). Note that these energy consumptions are
604 purely electrolytic. If the lamp consumption is added (i.e., the term $E_{\text{cell}}I$ in Eq. (6) is replaced by
605 the lamp power to determine $(\text{EC})_{\text{photo}}$,⁸² the EC_{TOC} values increase up to ~ 30 and ~ 36 kWh / g
606 TOC using BDD and DSA, respectively. In practice, natural sunlight can be used instead artificial
607 UVA light, thus avoiding the large contribution of $(\text{EC})_{\text{photo}}$.

608 In Figure 8a-d, the SEM images of GDEs prepared with both chitosan-based electrocatalysts
609 can be seen at two different magnifications. A good surface coverage of carbon cloth was obtained
610 with MC-C and N-MC-C, with no evidence of exposed carbon fibers. This is attributed to the good
611 spraying procedure, giving rise to several layers, followed by hot pressing. Some gaps can be
612 appreciated, which are better identified at the highest magnification (Figure 8b and 8d). These

613 macropores favored the mass transport rate of O₂ toward the catalytic sites.⁵ Amorphous particles
614 with dimensions of some microns appear in both types of GDE, as is typical for C-PTFE GDEs.
615 Some of them formed clusters wrapped around the fibers (Figure 8b). PTFE, recognized as the
616 white spots, was spread over the surface, conferring sufficient hydrophobicity to the materials.
617 Note that no flooding of the GDEs was observed during the trials of this work. The GDEs used in
618 PEF treatments with DSA described in Figure 7 were collected at the end of the electrolyses. The
619 corresponding SEM analysis (Figure 8a'-d') at the same magnifications mainly show smoother
620 surfaces without clear distinction of carbon particles and, more important, the obvious presence of
621 uncoated or partially coated fibers. This phenomenon was more evident in the cathode prepared
622 with MC-C, which can be directly explained from its aforementioned lower stability. Oxygen gas
623 is forced through its compact structure, causing a partial disintegration. However, this did not
624 impede to reuse all the GDEs for several consecutive runs without any substantial loss of
625 performance related to H₂O₂ production and degradation ability.

626 **Reaction products identified during the PEF treatment of acebutolol solutions.** In the
627 previous subsection, the TOC-time and MCE-time profiles have been justified on the basis of the
628 different reactivity of the reaction products. To demonstrate this, a 0.046 mM acebutolol solution
629 with 0.050 M Na₂SO₄ and 0.50 mM Fe²⁺ at pH 3.0 was treated by PEF for 60 min, using a cell
630 with the RuO₂-based anode connected to a GDE made with N-MC-C catalyst, at 30 mA/cm².
631 Samples collected at 10 and 60 min and analyzed by GC/MS revealed the presence of six aromatic
632 and three nitrogenated aliphatic compounds, whose characteristics are summarized in Table S3. In
633 Figure 9, the generation of these molecules is proposed considering the modification of parts R1,
634 R2, R3 and R4 distinguished in acebutolol. Some authors detected a product similar to **2**, but
635 hydroxylated in C1 position, when acebutolol solutions were treated by droplet-assisted
636 heterogeneous EF.⁸³ They also identified a structure close to **6** but keeping R1, and another one
637 like **7** but non-hydroxylated.

638 Most of these nine products were expected to be gradually hydroxylated.⁵ Therefore, the
639 evolution of the concentration of short-chain linear carboxylic acids during the same kind of PEF
640 treatment was investigated. Figure 10 evidences the trends of oxalic acid, whose peak was found
641 at retention time of 7.0 min, and oxamic acid, which appeared at 9.8 min. The former attained a
642 maximum concentration of 1.75 mg/L at 30 min, quickly decaying to finally disappear at 120 min.
643 Oxalic acid is a typical end organic product from aromatic pollutants, which is the reason that
644 explains that it was the main accumulated carboxylic acid.⁵¹ The high photoactivity of its
645 complexed form under UVA light according to reaction (14) justifies its total degradation in PEF.
646 In contrast, oxamic was only formed from larger N-structures, reaching a lower content of 1.05
647 mg/L at 60 min. Moreover, since it formed much less photoactive complexes,⁵¹ it persisted in the
648 solution during the entire electrolysis. The presence of both acids as acebutolol is degraded has
649 also been reported elsewhere.⁸³

650 The inorganic ions generated in PEF treatment from the heteroatoms contained in the drug
651 were determined as well. A more concentrated acebutolol solution (i.e., 0.092 mM) was employed
652 in order to facilitate the quantification. Neither nitrite nor nitrate were identified, being NH_4^+ the
653 only dissolved nitrogenated ion detected, thus corroborating the validity of mineralization reaction
654 (15). The ammonium content increased all the time, as the organic N-structures were degraded,
655 attaining 1.43 mg/L at 360 min. This cation was also the main nitrogenated ion found during the
656 degradation of other N-aromatic pollutants by Fenton-based electrochemical processes.⁷⁰ A
657 nitrogen balance after normalizing the acebutolol content suggests that the treated solutions
658 contained ~50% of initial dissolved nitrogen, most of it corresponding to NH_4^+ . This means that
659 some N-containing gases were released during the PEF treatment of acebutolol.

660 CONCLUSIONS

661 This work reports the successful synthesis and characterization of a potentially cheaper and more
662 eco-friendly material to electrogenerate sufficient amounts of H₂O₂ for its direct application in
663 Fenton-based electrochemical water treatment. Chitosan was considered as raw material for the
664 preparation of N-doped carbon. It was observed that the addition of 1,10-phenanthroline as co-
665 reactant during the pyrolysis treatments conferred better textural properties and surface N-
666 functionalization than the sole chitosan. LSV at RRDE highlighted the higher catalytic activity of
667 N-MC-C, in terms of $E_{1/2}$. GDEs prepared with N-MC-C showed H₂O₂ yields in the same range
668 attainable with commercial GDE. The degradation of 0.046 mM acebutolol solutions was made
669 with the N-MC-C-derived GDE coupled to a RuO₂-based anode in an electrochemical cell at pH
670 3.0. The PEF assays in such cell resulted in a fast and complete drug removal. The use of BDD
671 instead of DSA had a positive influence on the decontamination, since the almost complete TOC
672 removal was achieved at the end of the electrolysis. GC/MS analysis clarified the main aromatic
673 and aliphatic products formed during the mineralization process. Short-chain linear carboxylic
674 acids such as oxalic and oxamic were generated as final organic compounds, whereas NH₄⁺ was
675 the main ion.

676 ASSOCIATED CONTENT

677 Supporting Information (SI)

678 Figures S1-S5.

679 Tables S1-S3.

680 The Supporting Information is available free of charge on the ACS Publications website.

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966

Figure captions

Figure 1. Different stages during the synthesis of MC-C catalysts: (a) chitosan viscous solution, (b) hydrogel coagulation (c) chitosan hydrogel, (d) hydrogel after freeze drying, (e) dried hydrogel after pyrolysis at 400 °C, (f) first pyrolysis product after milling, (g) product after pyrolysis at 900 °C.

Figure 2. TEM images of (a-c) MC-C and (d-f) N-MC-C catalysts.

Figure 3. (a) Nitrogen adsorption/desorption isotherms of MC-C and N-MC-C catalysts; (b) Raman spectra for both materials; C 1s XPS core-level spectra for (c) MC-C and (d) N-MC-C catalysts; N 1s XPS core-level spectra for (e) MC-C and (f) N-MC-C. In plot (c-f), deconvolution signals are also shown.

Figure 4. Electrochemical data for the investigated materials, drop-cast on glassy carbon (catalyst loading: 0.3 mg/cm²), in Ar-purged and O₂-saturated 0.0005 M H₂SO₄ + 0.050 M Na₂SO₄ electrolyte. Cyclic voltammograms using (a) MC-C and (b) N-MC-C, at $v_{\text{scan}} = 20$ mV/s. Linear sweep voltammograms at RRDE, with at $v_{\text{scan}} = 10$ mV/s and $\omega = 1600$ rpm, using (c) MC-C and (d) N-MC-C. (c', d') Ring current, at $E_{\text{ring}} = 1.50$ V vs. RHE, and (c'', d'') H₂O₂ selectivity during the linear sweep voltammetry in O₂-saturated electrolyte, using (c', c'') MC-C and (d', d'') N-MC-C.

Figure 5. Evolution of H₂O₂ concentration during the electrolysis of 150 mL of 0.050 M Na₂SO₄ solutions at pH 3.0 and 25 °C, using a cell with the MC-C or N-MC-C cathode and a DSA plate as the anode at different current densities and air flow rate of 36 L/h. Process: (a) EO-H₂O₂, and (b) EF with 0.50 mM Fe²⁺ and PEF with 0.50 mM Fe²⁺ and a 6 W UVA lamp.

Figure 6. Time course of acebutolol concentration during the treatment of 150 mL of solutions with 0.046 mM drug and 0.050 M Na₂SO₄ at pH 3.0 and 25 °C, using a cell with the MC-C or N-

MC-C cathode at a current density of 30 mA/cm² and air flow rate of 36 L/h. Process (a) EO-H₂O₂ with DSA, and (b) PEF with 0.50 mM Fe²⁺, a 6 W UVA lamp and DSA or BDD as the anode. In (b), the inset panel presents the corresponding pseudo-first-order kinetic analysis.

Figure 7. Change of normalized TOC with electrolysis time for the PEF trials shown in Fig. 6b.

Figure 8. SEM images of GDEs prepared with the chitosan-based electrocatalysts. (a-d) Fresh and (a'-d') used. Magnification: (a,a',c,c') 100× and (b,b',d,d') 500×. The used GDEs were those collected at the end of the PEF treatments with DSA described in Fig.7.

Figure 9. Primary products of acebutolol during PEF treatment with chitosan-derived electrocatalysts.

Figure 10. Evolution of the concentration of aliphatic carboxylic acids during the PEF treatment of 0.046 mM acebutolol solutions with 0.050 M Na₂SO₄ and 0.50 mM Fe²⁺, as well as NH₄⁺ content accumulated during an analogous treatment but using 0.092 mM drug and 0.020 M Na₂SO₄ instead. Conditions: cell with N-MC-C cathode (fed with air at 36 L/h) and the RuO₂-based anode, containing 150 mL of solutions at pH 3.0 and 25 °C exposed to a 6 W UVA lamp and electrolyzed at 30 mA/cm².

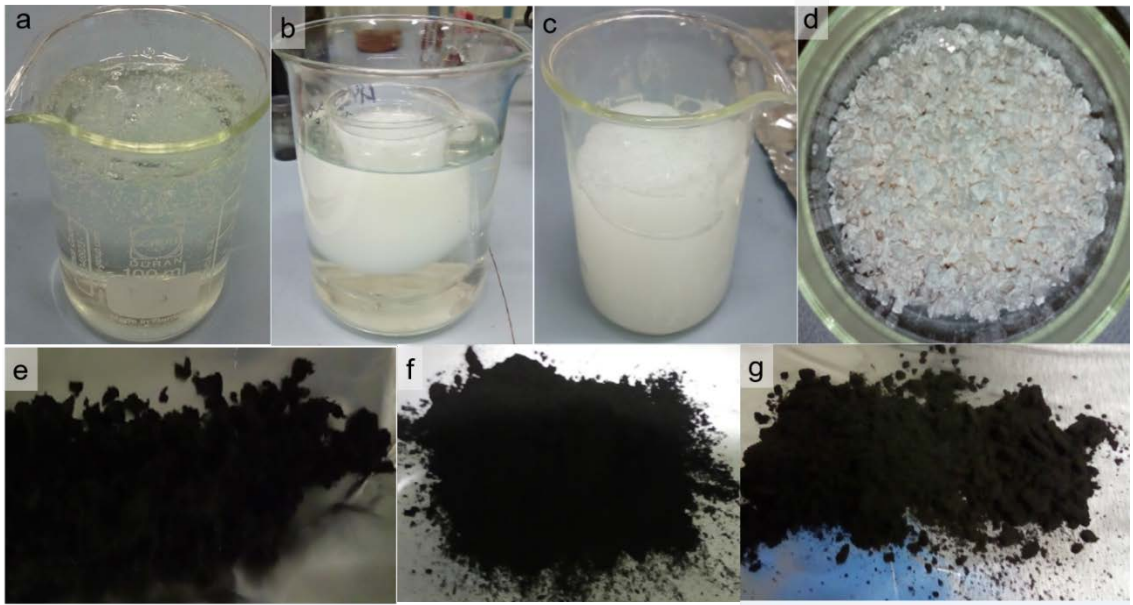


Figure 1

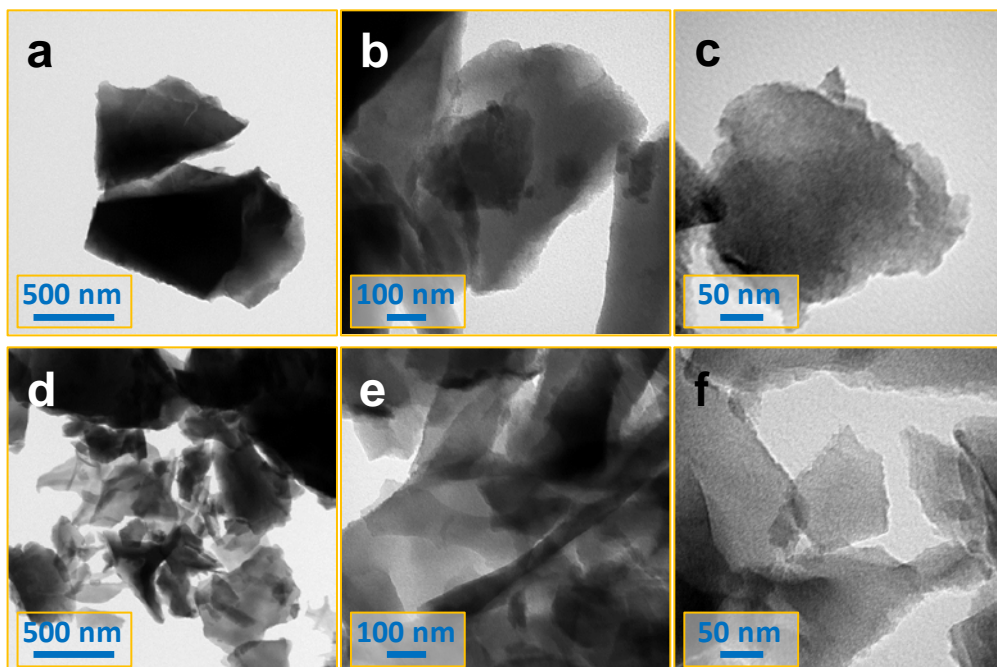


Figure 2

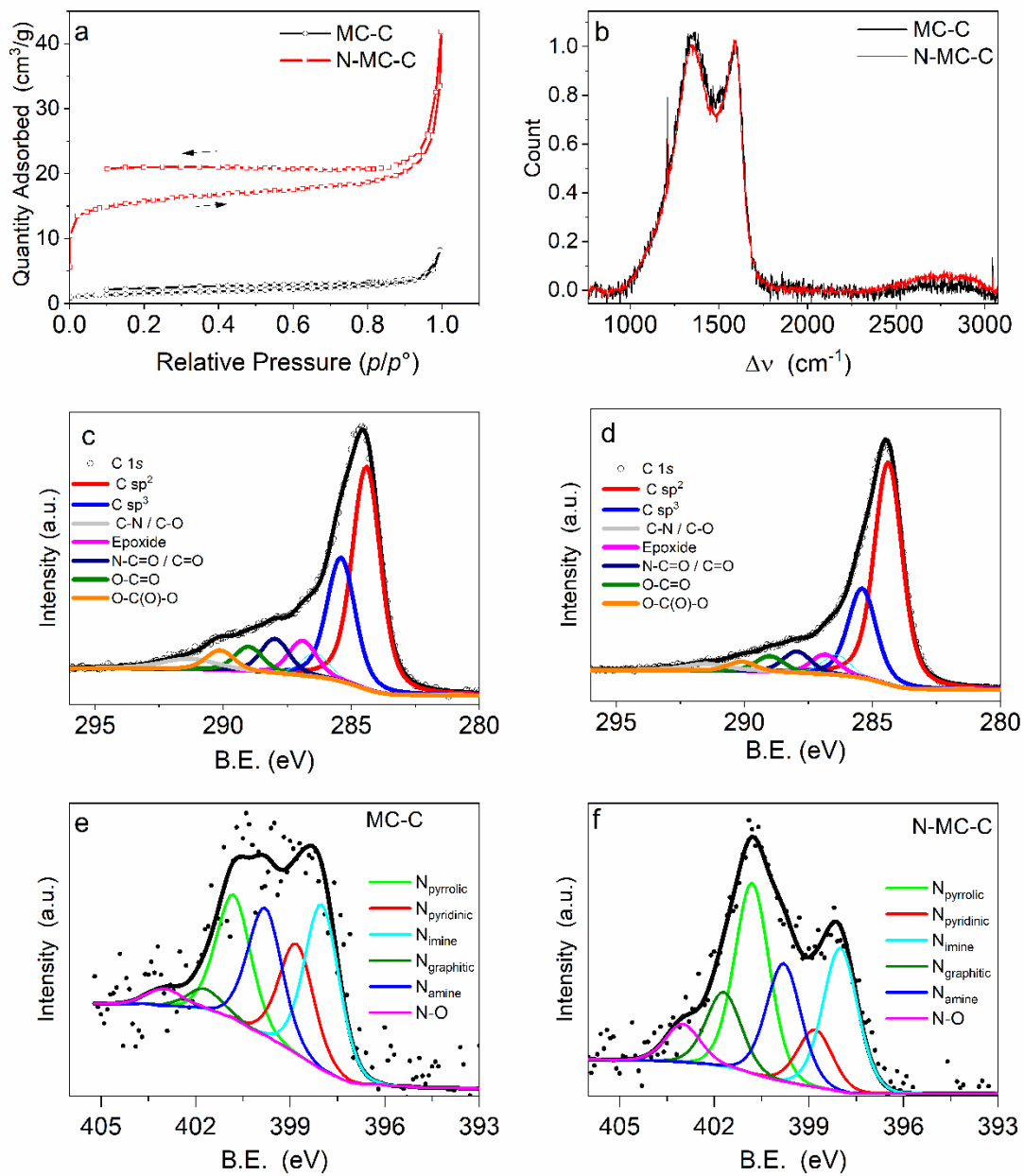


Figure 3

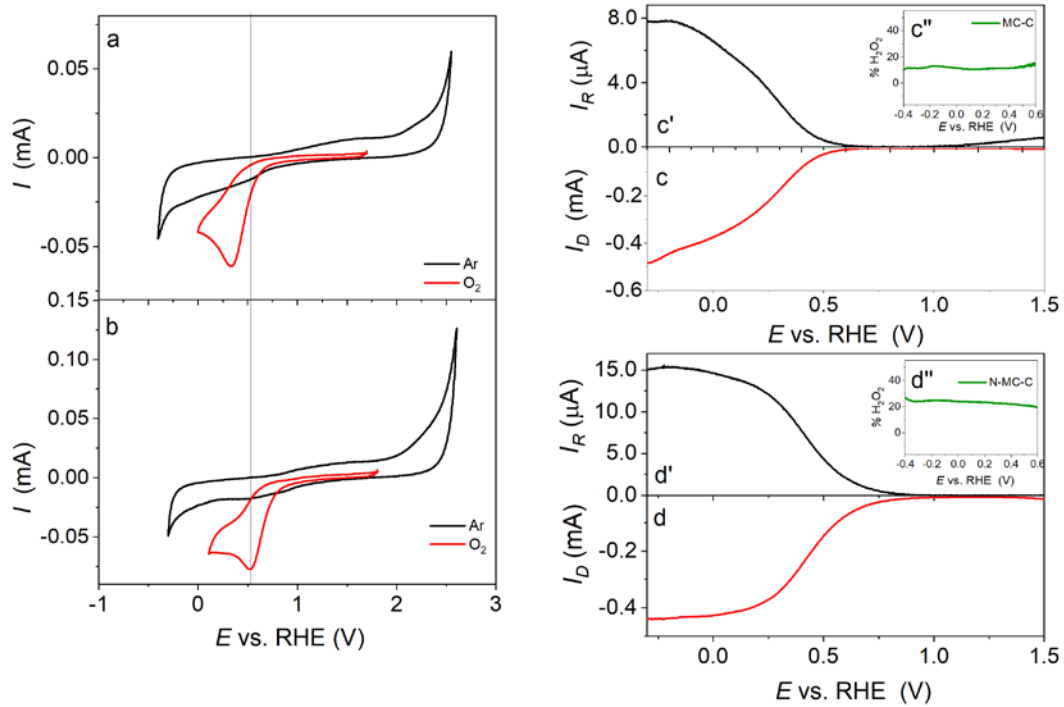


Figure 4

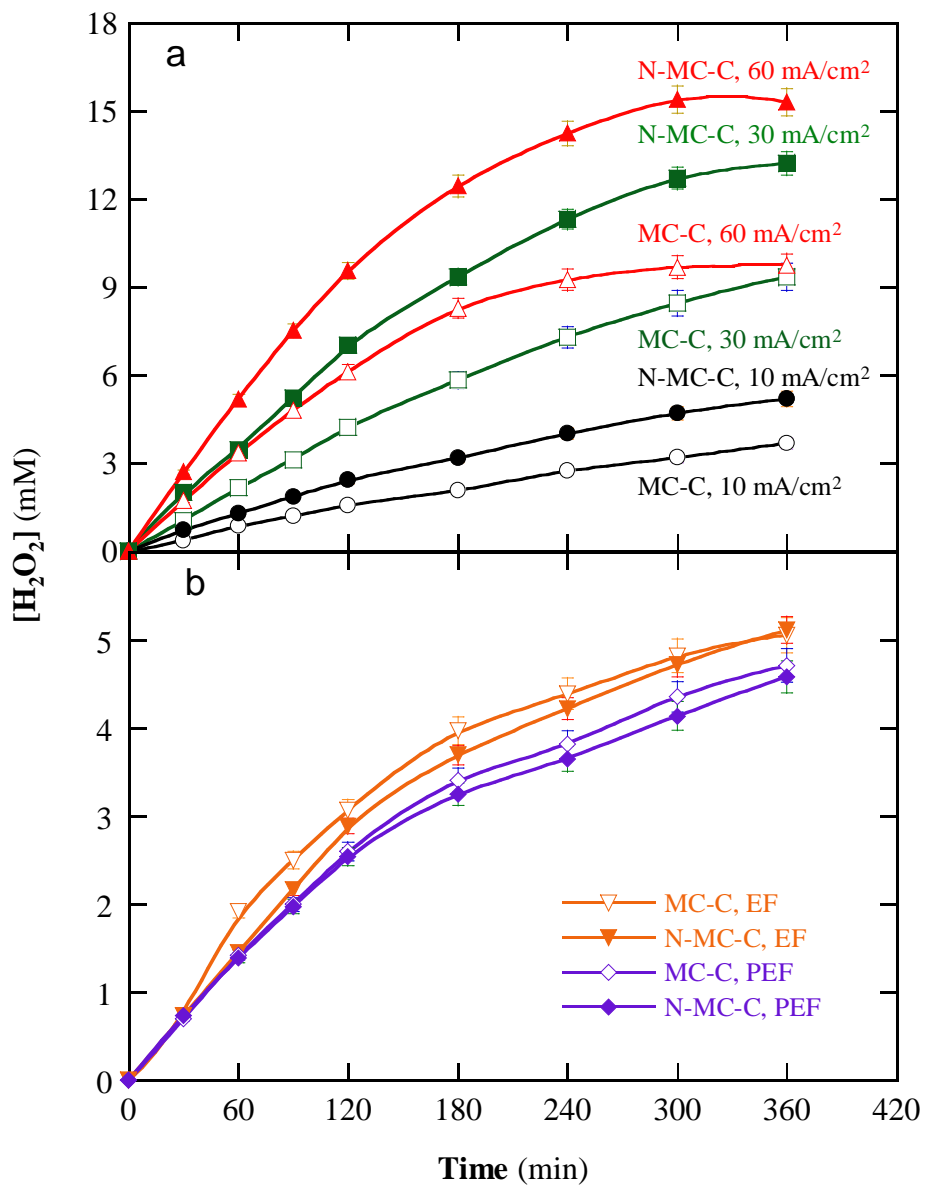


Figure 5

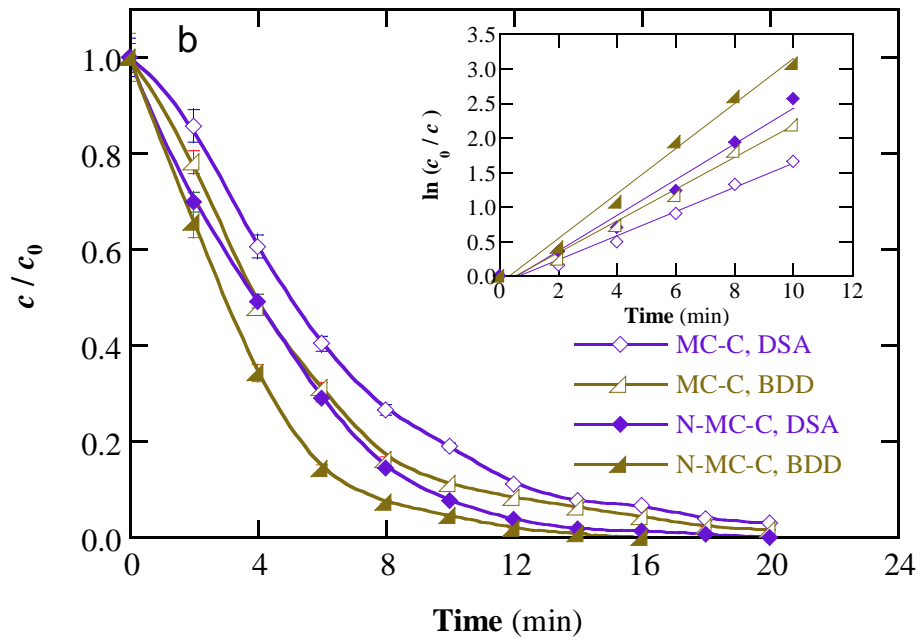
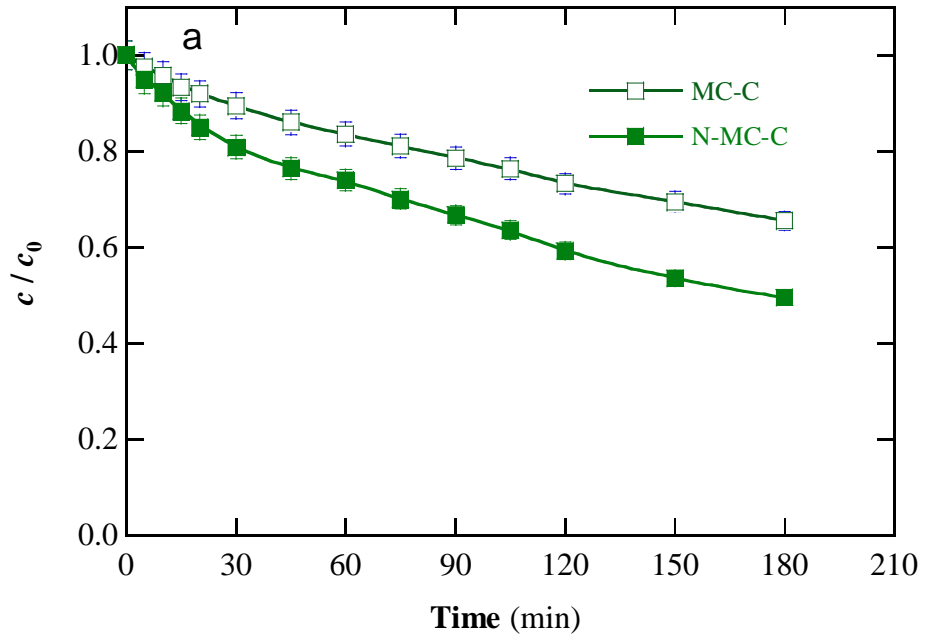


Figure 6

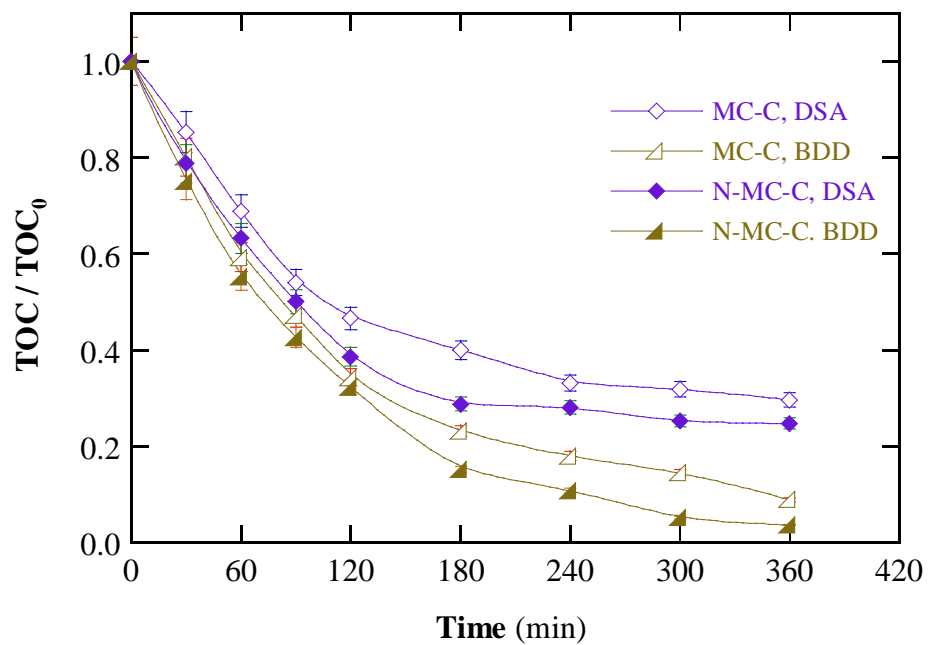
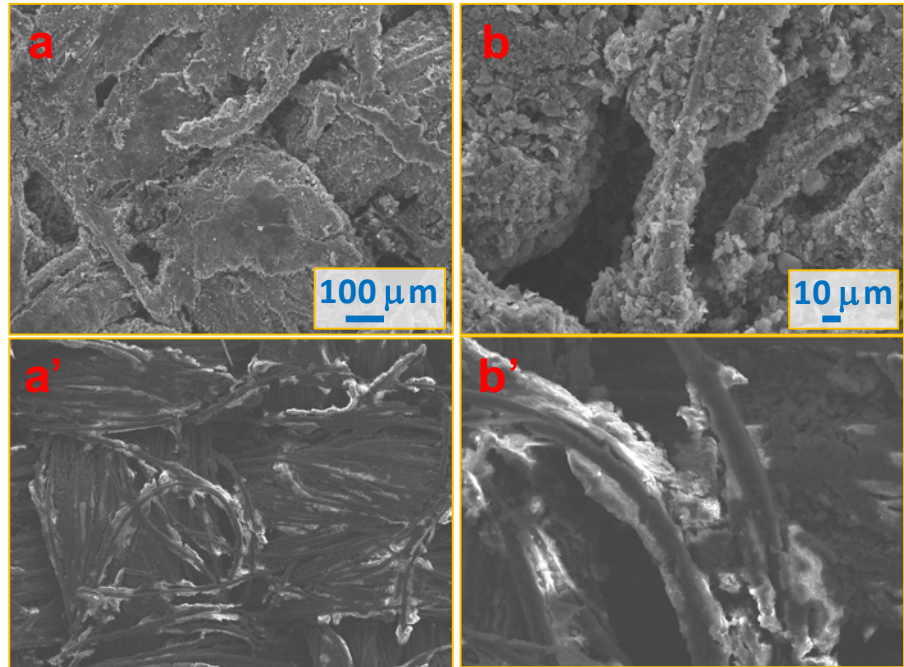


Figure 7

MC-C



N-MC-C

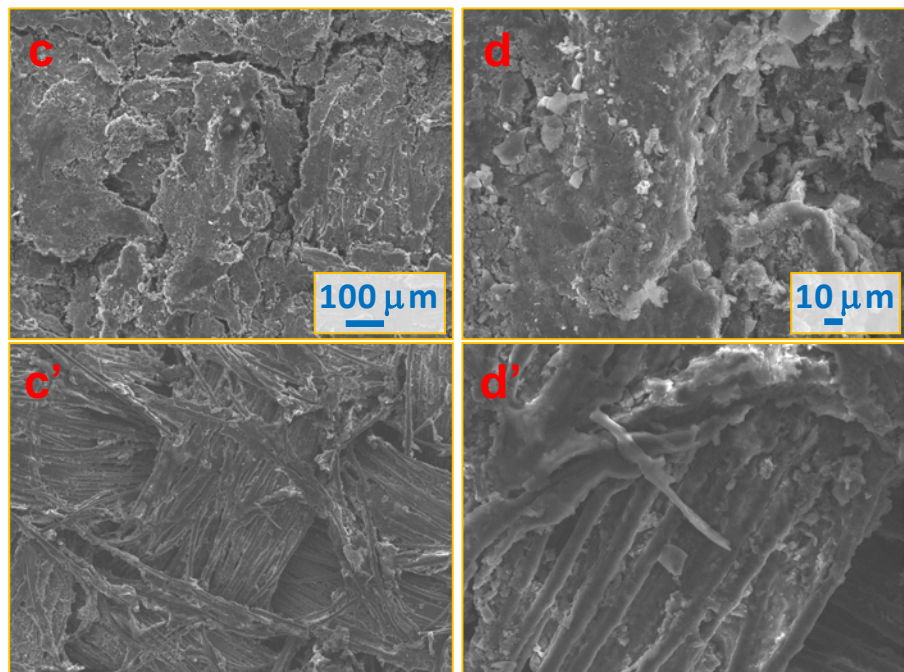


Figure 8

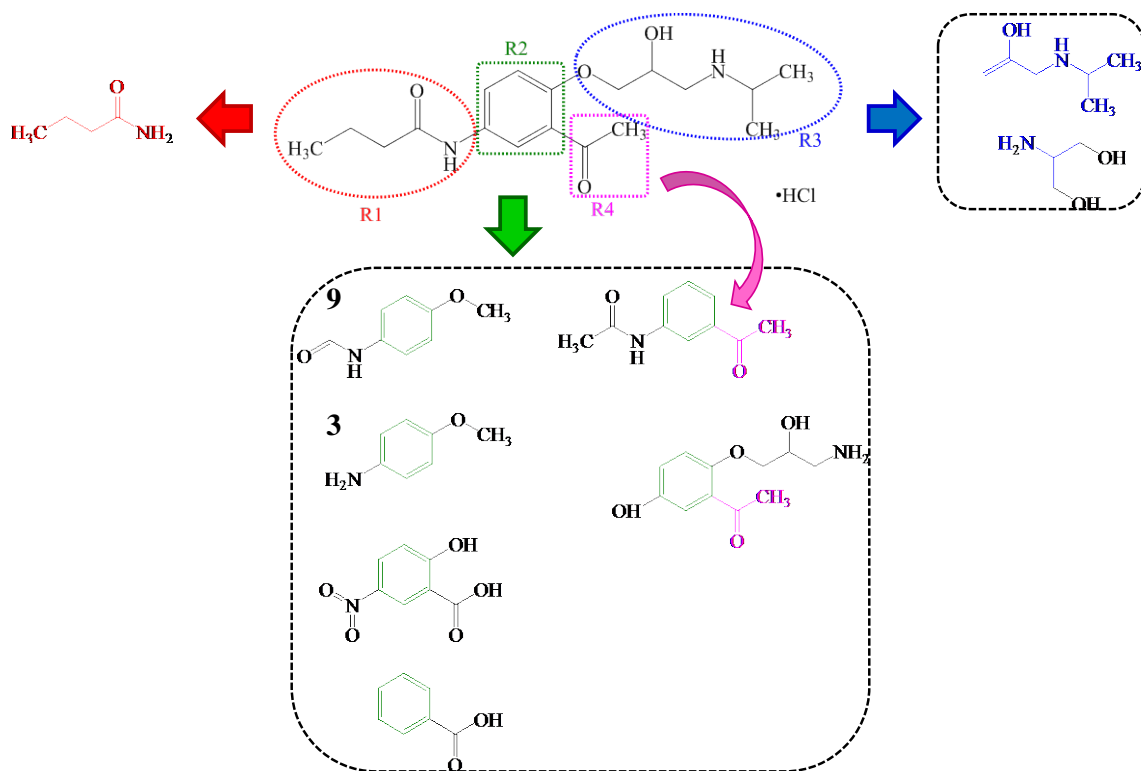


Figure 9

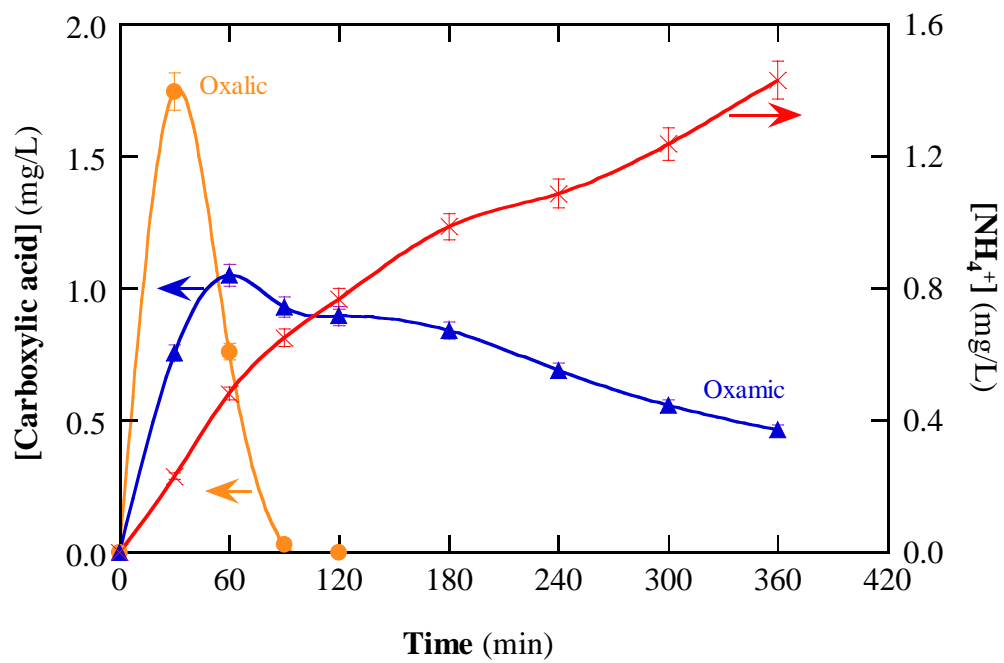


Figure 10

Table 1

Chemical and textural properties of activated carbons.

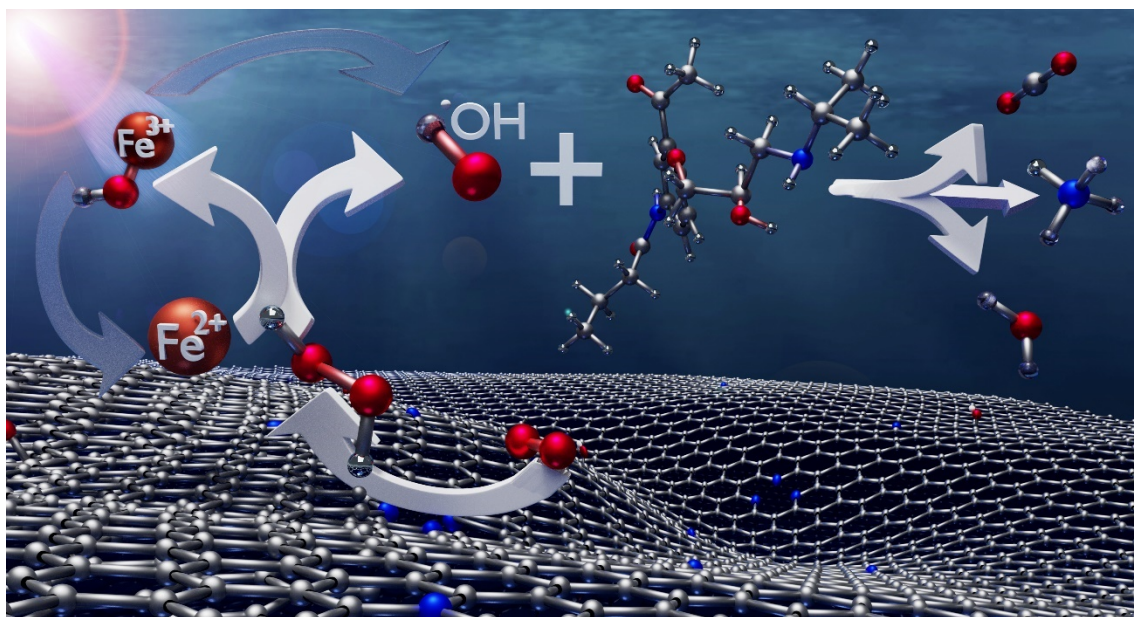
Sample	C^a (%)	H^a (%)	N^a (%)	A_{BET}^b (m ² /g)	V_{TOT}^b (cm ³ /g)	V_μ^b (cm ³ /g)	V_{meso}^b (cm ³ /g)	V_{DFT}^b (cm ³ /g)
MC-C	76.44	0.81	6.81	6	0.013	0.002	0.005	0.007
N-MC-C	79.39	1.73	4.62	63	0.065	0.021	0.017	0.038

Sample	C 1s^c (at %)	O 1s^c (at %)	N 1s^c (at %)	N_{imine}^c (%)	N_{pyridinic}^c (%)	N_{aminic}^c (%)	N_{pyrrolic}^c (%)	N_{graphitic}^c (%)	N-O^c (%)
MC-C	78.5	19.1	2.4	29.3	19.5	22.7	22.0	3.5	3.0
N-MC-C	81.6	15.3	3.1	23.1	9.2	19.0	30.7	12	6.1

^aElemental analysis determined by CHN analyser.^bTexture properties determined from nitrogen adsorption-desorption isotherms.^cElemental analysis expressed in atomic percentage and nitrogen functionalities determined from XPS analysis and N 1s spectra deconvolution.

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A more sustainable H_2O_2 electrosynthesis from O_2 reduction, aiming at $\cdot\text{OH}$ -mediated water treatment, by employing an air-diffusion cathode with biomass-derived N-doped carbon